

# MODELING LIQUID METALS and BULK METALLIC GLASSES

Jean BELLISSARD

*Georgia Institute of Technology, Atlanta  
School of Mathematics & School of Physics  
e-mail: jeanbel@math.gatech.edu*

Sponsoring



*This material is based upon work supported by the National Science Foundation*

*Grant No. DMS-1160962*



*Any opinions, findings, and conclusions or recommendations expressed in this material are those of the author(s) and do not necessarily reflect the views of the National Science Foundation.*

# Teaching, Counseling, Support

T. EGAMI, (*JINS*, Oak Ridge & U. Tennessee, Knoxville)

# Main References

- T. EGAMI, *Atomic Level Stress*, Prog. Mat. Sci., **56**, (2011), 637-653.
- W. H. WANG, C. DONG, C. H. SHEK, *Bulk Metallic Glasses*, Mater. Sci. Eng. Rep., **44**, (2004), 45-89.
- M. MILLER, P. LIAW, Eds., *Bulk Metallic Glasses: An Overview*, Springer, (2007).
- H. S. CHEN and D. TURNBULL, *J. Chem. Phys.*, **48**, 2560-2571, (1968).
- MORREL H. COHEN & G. S. GREY, *Phys. Rev. B*, **20**, 1077-1098, (1979).
- D. J. PLAZEK, J. H. MAGILL, *J. Chem. Phys.*, **45**, 3757, (1967); J. H. MAGILL, *ibid.* **47**, 2802, (1967).
- J. HAFNER, *Phys. Rev. B*, **27**, 678-695 (1983).
- M. L. FALK, J. S. LANGER, L. PECHENIK, *Phys. Rev. E*, **70**, 011507, (2004).
- J. D. BERNAL, *the Structure of Liquids*, Proc. Roy. Soc., **A280**, (1964), 299-322.
- D. B. MIRACLE, *A structural model for metallic glasses*, Nature Mat., **3**, (2004), 697-702.
- D. B. MIRACLE, W. S. SANDERS, N. SENKOV, *Phil. Mag.*, **83**, (2003), 2409-2428.
- D. MA, A. D. STOICA, X.-L. WANG, *Nature Mat.*, **8**, (2009), 30-34.
- T. EGAMI, D. SROLOVITZ, *J. Phys. F*, **12**, 2141-2163 (1982).
- S.-P. CHEN, T. EGAMI, V. VITEK, *Phys. Rev. B*, **37**, 2440-2449, (1988).
- T. EGAMI, S. J. POON, Z. ZHANG, V. KEPPENS, *Phys. Rev. B*, **76**, 024203, (2007).

# Content

1. Metal Liquids and Glasses
2. Cluster Models
3. Delone Graphs
4. The Anankeon Theory
5. Towards a Dissipative Dynamics

# I - Metal Liquids and Glasses

# Bulk Metallic Glasses

## 1. Examples *(Ma, Stoica, Wang, Nat. Mat. '08)*

- $\text{Zr}_x\text{Cu}_{1-x}$      $\text{Zr}_x\text{Fe}_{1-x}$      $\text{Zr}_x\text{Ni}_{1-x}$
- $\text{Cu}_{46}\text{Zr}_{47-x}\text{Al}_7\text{Y}_x$      $\text{Mg}_{60}\text{Cu}_{30}\text{Y}_{10}$

## 2. Properties *(Hufnagel web page, John Hopkins)*

- High *Glass Forming Ability* (GFA)
- High *Strength*, comparable or larger than steel
- Superior *Elastic limit*
- High *Wear* and *Corrosion* resistance
- *Brittleness* and *Fatigue* failure

# Bulk Metallic Glasses

Applications *(Liquidmetal Technology [www.liquidmetal.com](http://www.liquidmetal.com))*

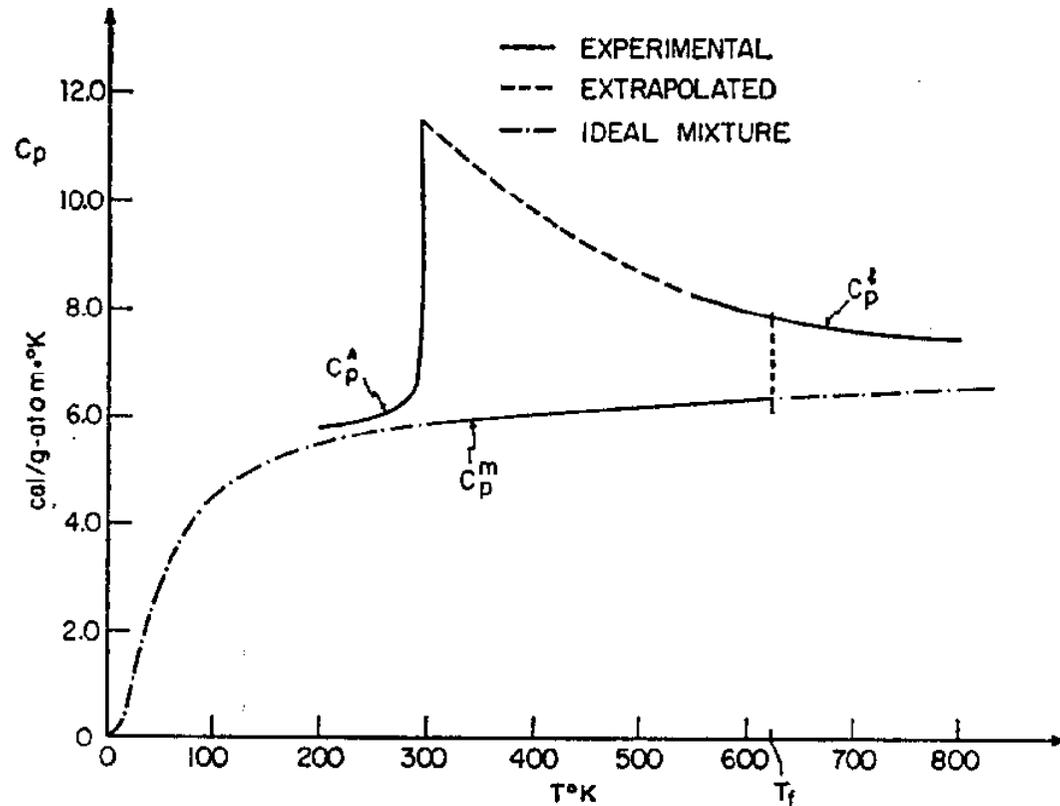
- *Orthopedic implants* and medical Instruments
- Material for *military components*
- Sport items, *golf clubs, tennis rackets, ski, snowboard, ...*



Pieces of Titanium-Based Structural  
Metallic-Glass Composites

*(Johnson's group, Caltech, 2008)*

# Bulk Metallic Glasses

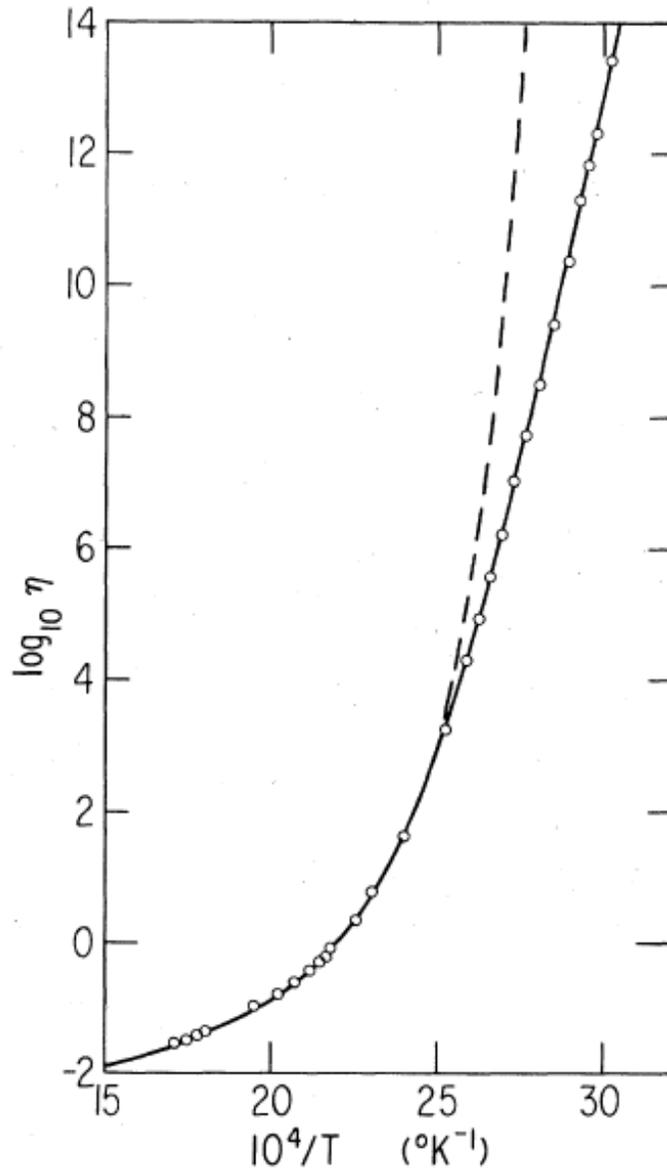


Smoothed values of specific heats of  $Au_{.77}Ge_{.136}Si_{.094}$  signaling a glass-liquid transition

"A" designates the amorphous state  
"m" designates the mixture  
"l" designates the liquid

*taken from*  
H. S. CHEN and D. TURNBULL, *J. Chem. Phys.*,  
48, 2560-2571, (1968)

# Bulk Metallic Glasses



Viscosity vs temperature for tri-anaphthylbenzene, with fits coming from the *free volume theory*

*Solid curve* fit from [1] below

*Dashed curve*: fit from [1] with a simplified theory

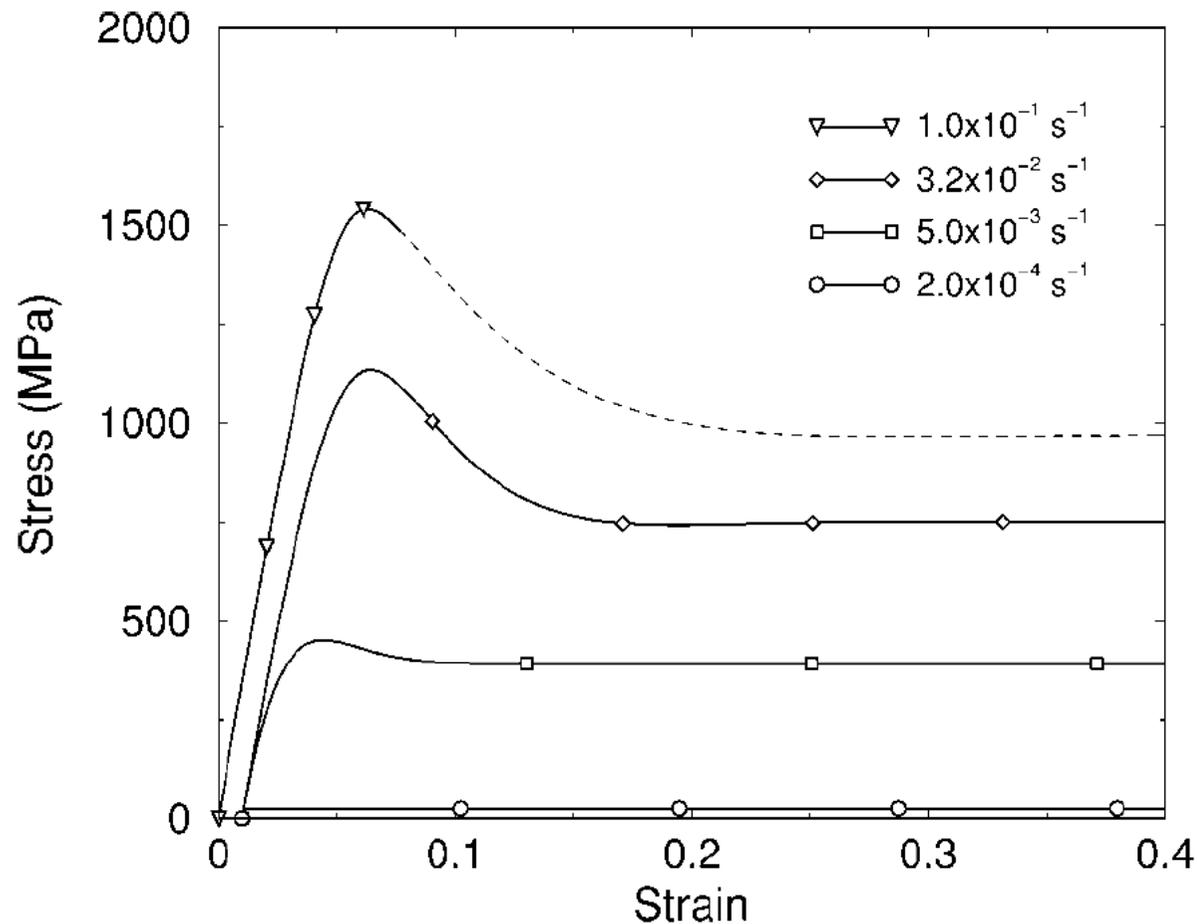
*Circles*: data from [2] below

*taken from*

[1] MORREL H. COHEN & G. S. GREY, *Phys. Rev. B*, **20**, 1077-1098, (1979)

[2] D. J. PLAZEK and J. H. MAGILL, *J. Chem. Phys.*, **45**, 3757, (1967); J. H. MAGILL, *ibid.* **47**, 2802, (1967)

# Bulk Metallic Glasses



Theoretical curves of tensile stress versus strain for the bulk metallic glass using the *STZ theory*



at several different strain rates as shown. The temperature is  $T=643 \text{ K}$ .

*For clarity, all but the first of these curves have been displaced by the same amount along the strain axis.*

*taken from*

[1] M. L. FALK, J. S. LANGER & L. PECHENIK, *Phys. Rev. E*, **70**, 011507, (2004)

# Bulk Metallic Glasses

1. No *structural difference* between liquid and glass. No sharp discontinuity of equilibrium variables
2. The *time scales change sharply* from liquid to glass. The glass transition temperature is defined by a conventional time scale beyond which the dynamics is hard to observe.

## II - Cluster Models

# Cluster Models

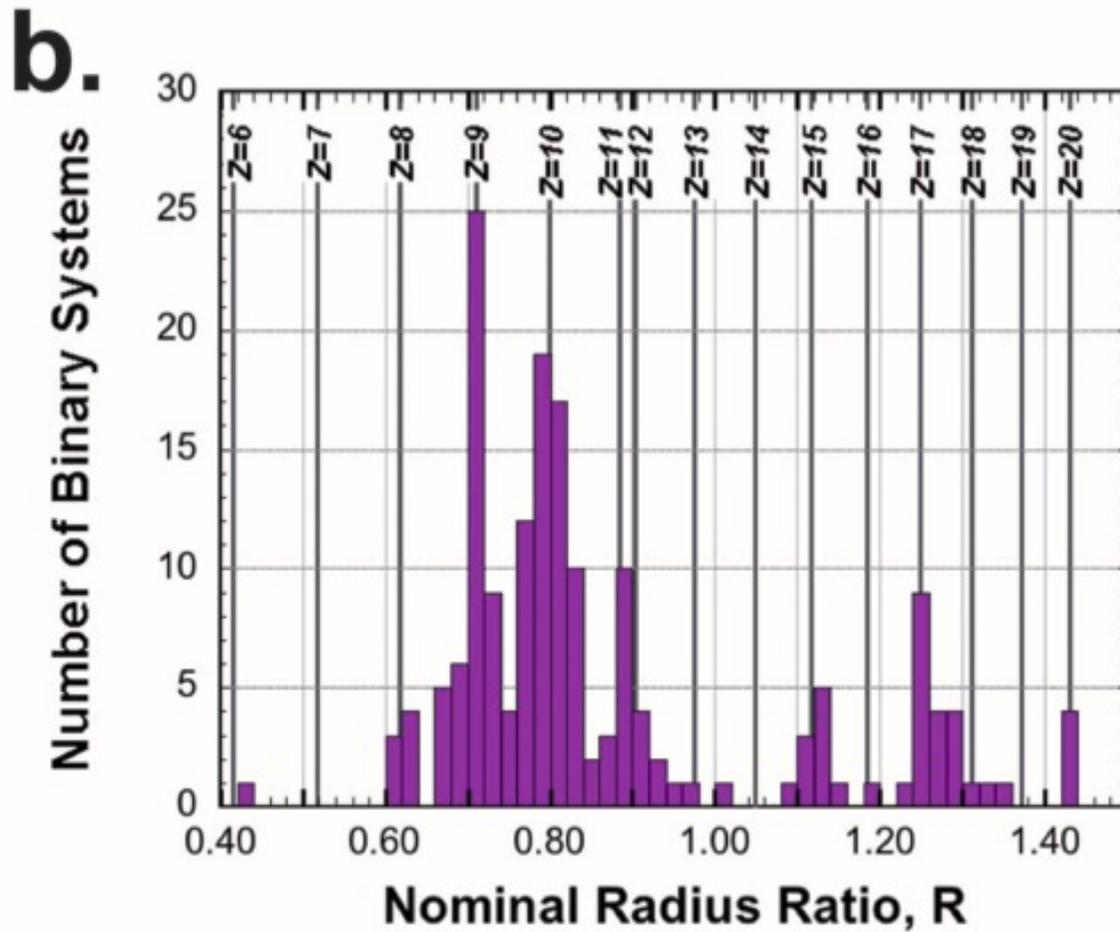
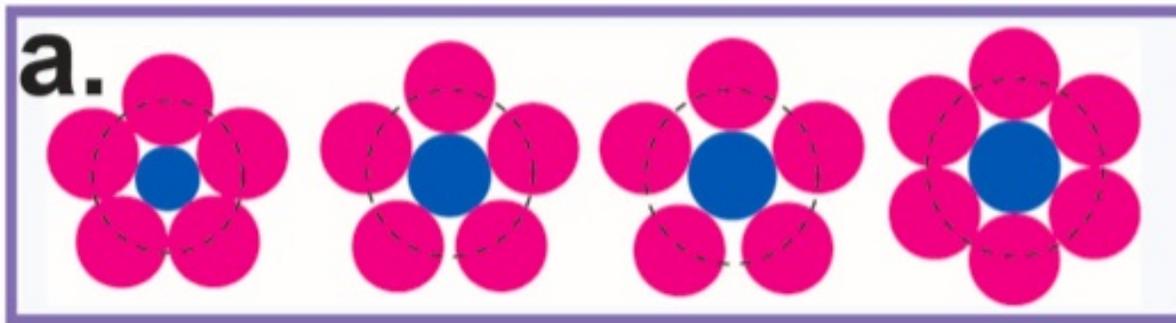
J. D. BERNAL, *The Structure of Liquids*, Proc. Roy. Soc., **A280**, (1964), 299-322.

T. EGAMI, Y. WASEDA, *Atomic size effect on the formability of metallic glasses*, J. Non Cryst. Sol., **64**, (1984), 113-134.

D. B. MIRACLE, W. S. SANDERS, N. SENKOV, Phil. Mag., **83**, (2003), 2409-2428.

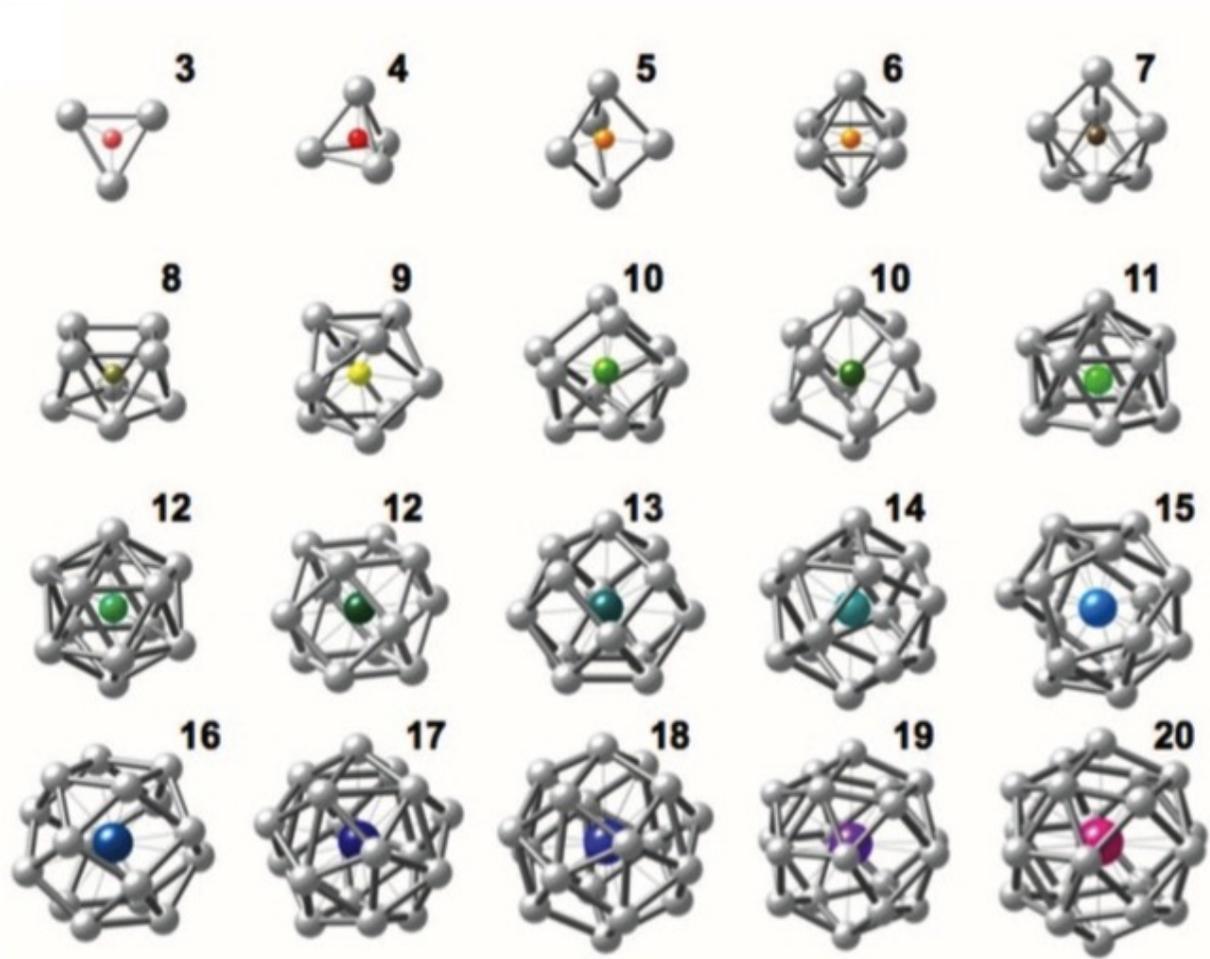
D. B. MIRACLE, *A structural model for metallic glasses*, Nature Mat., **3**, (2004), 697-702.

1. Cluster models are based on *densest packing* distributions of hard spheres.
2. A *cluster* is formed from one solute atom and a layer of solvent atoms.
3. At larger scales cluster behave like new particles with almost spherical shape.



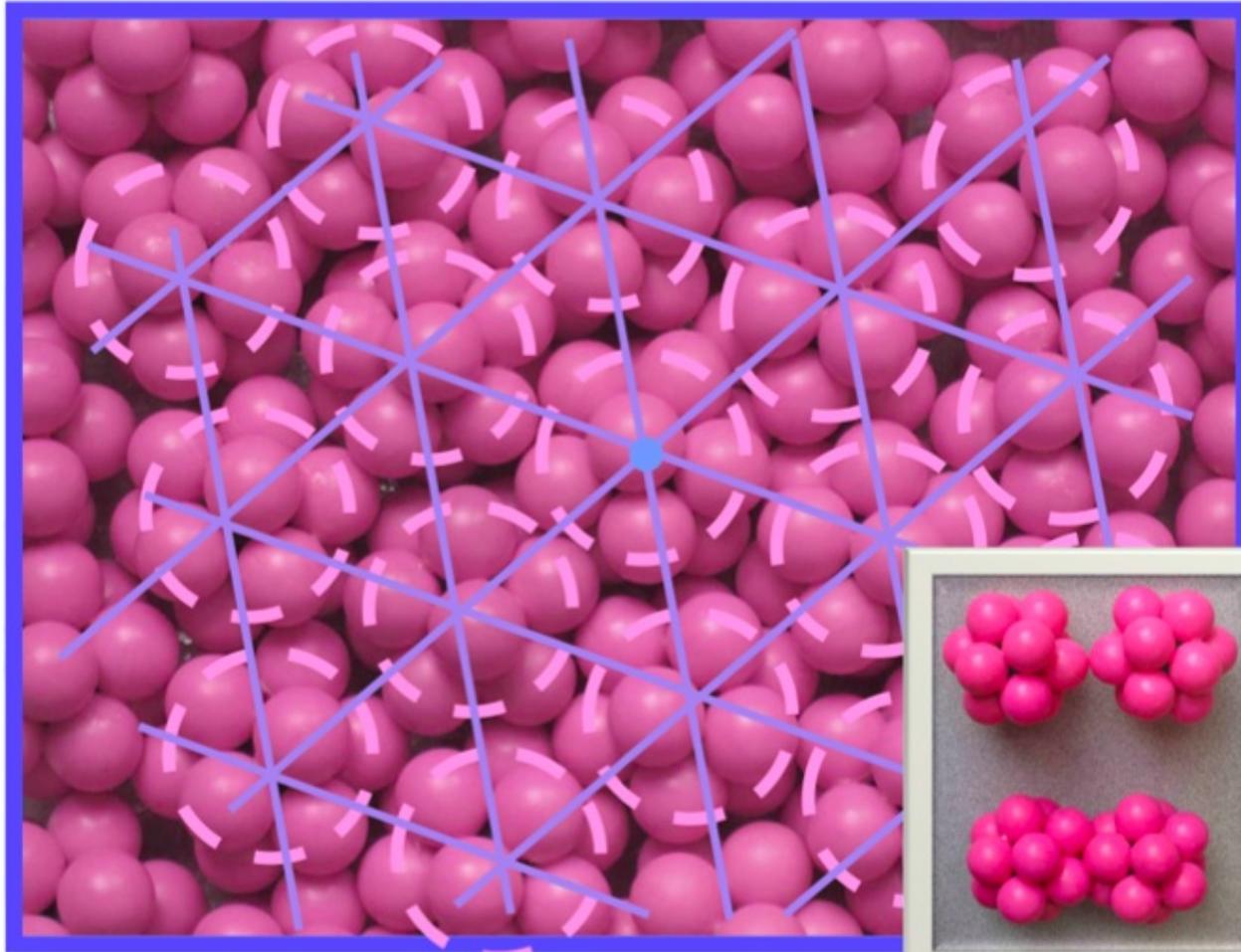
2D-clusters

*(T. Egami, Y. Waseda, '84)*  
*(D. Miracle, et al. '04)*



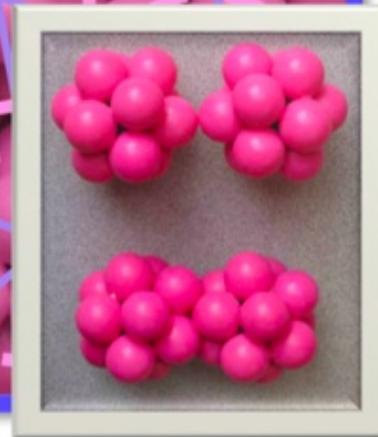
### 3D-clusters

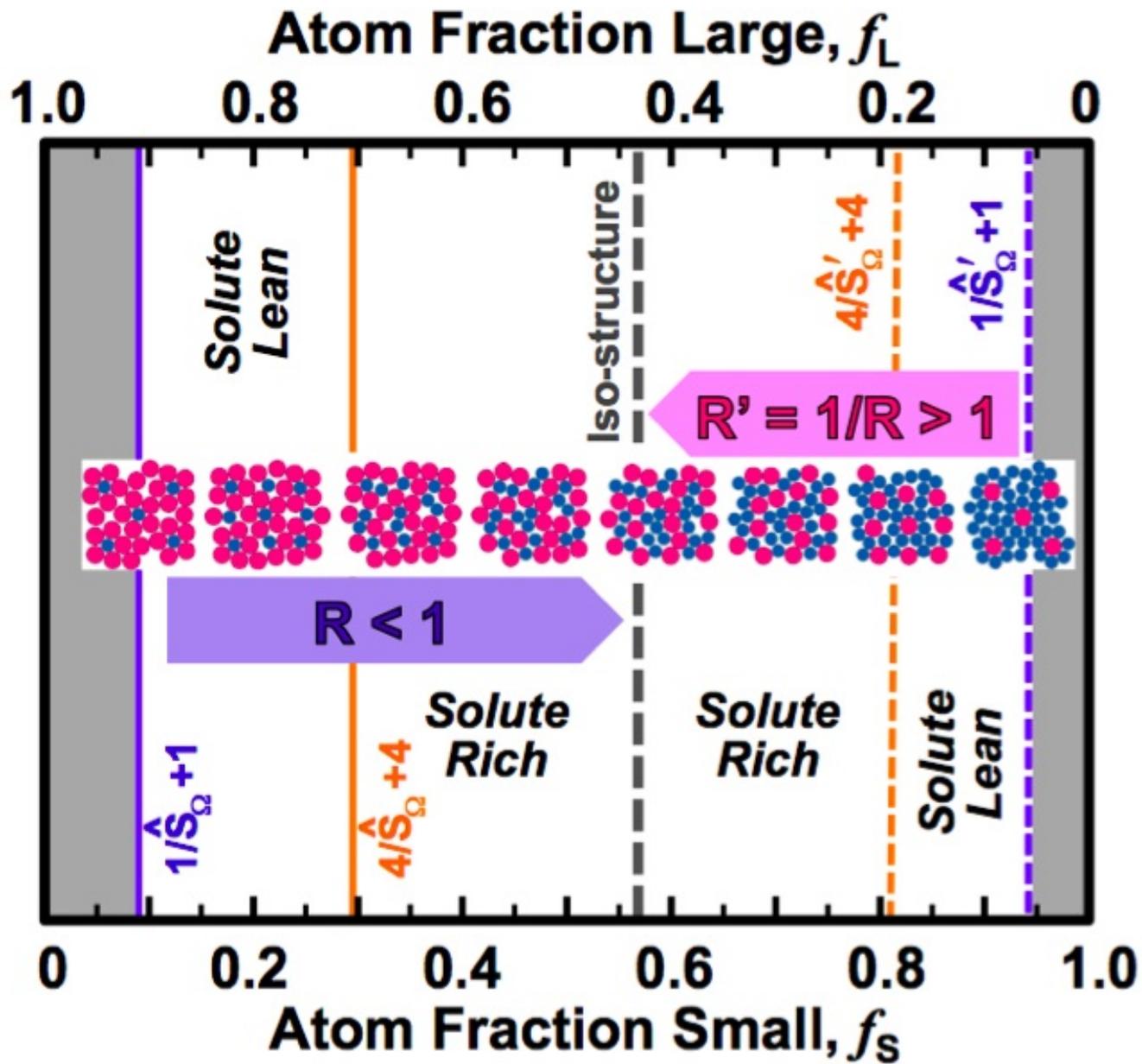
*(D. Miracle, et al. '04)*



## Packing Clusters

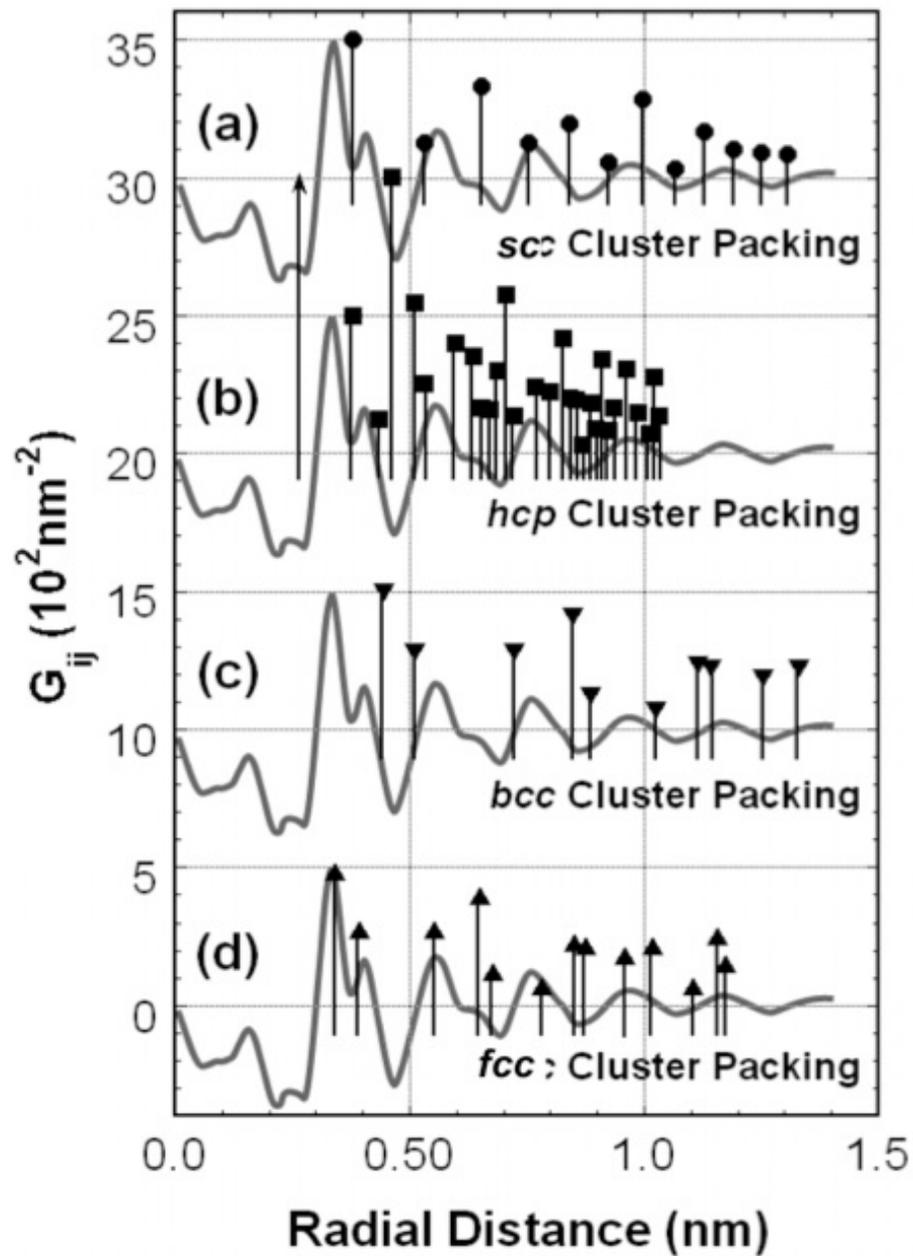
*(D. Miracle, et al. '04)*





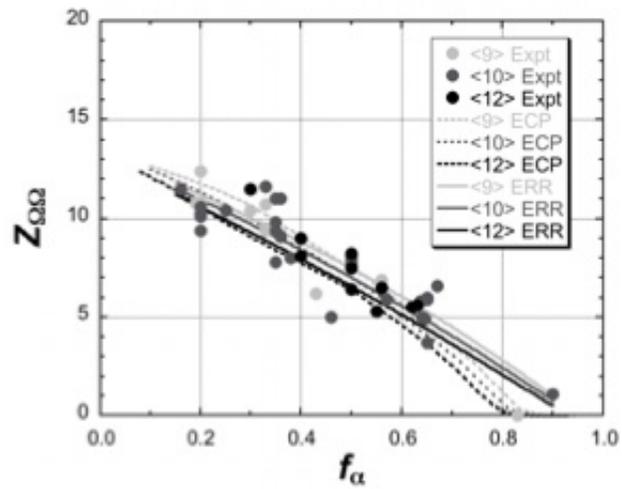
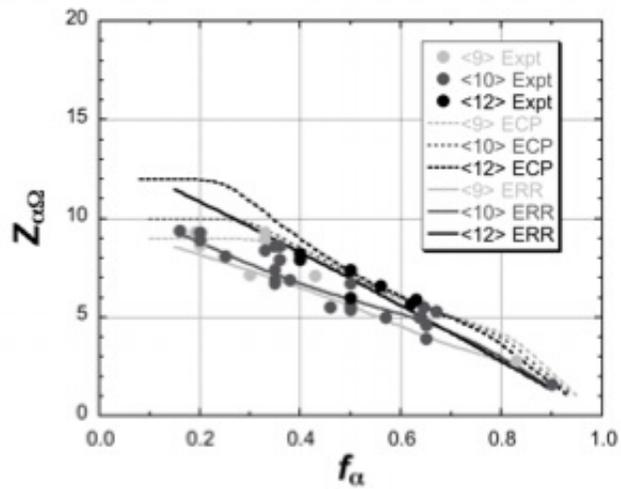
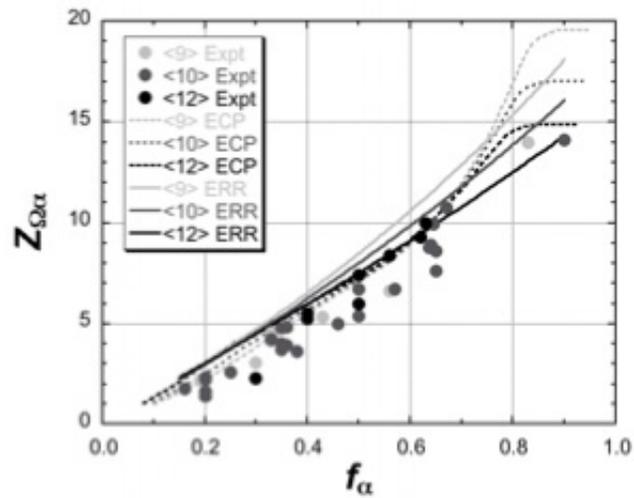
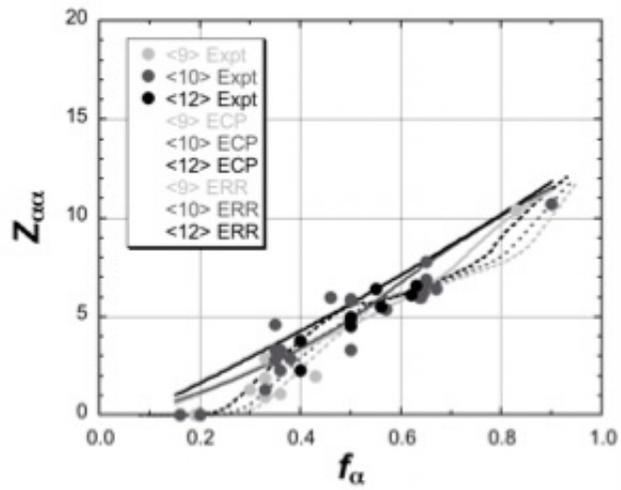
Clusters Evolution

(D. Miracle, et al. '04)



Comparison between cluster packing and Correlation for various crystals

*(D. Miracle, et al. '04)*



Measured (filled circles) and predicted (lines) partial coordination numbers.

*(D. Miracle, et al. '04)*

# Diffraction

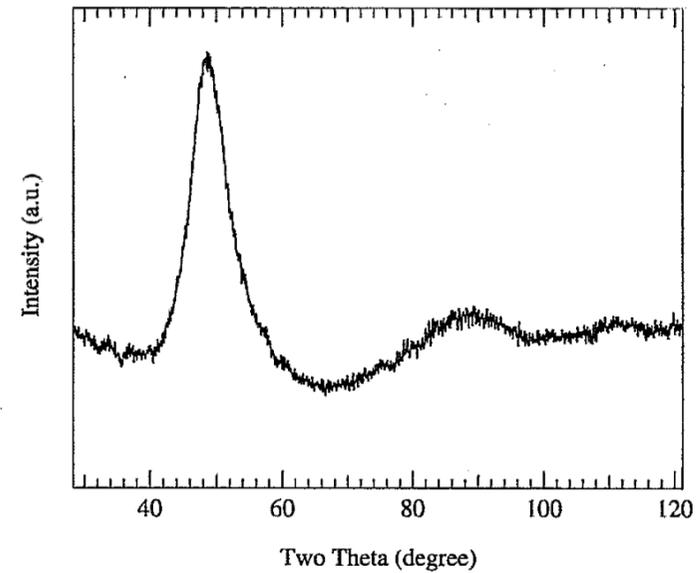
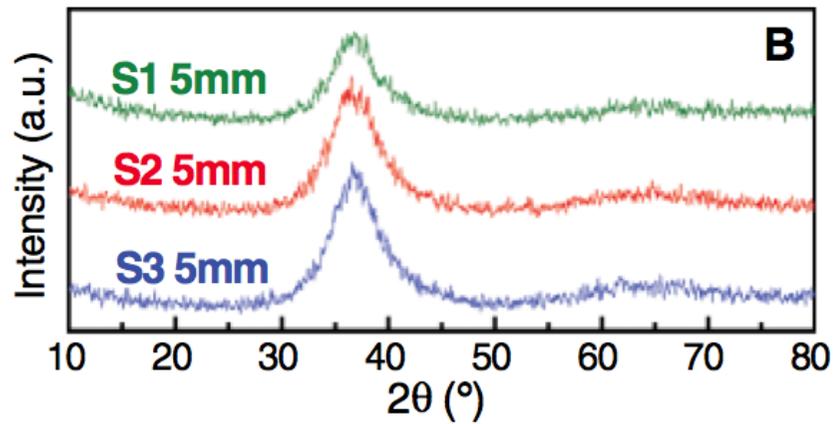


FIG. 3. X-ray-diffraction pattern (Co  $K\alpha$  radiation) taken from the cross-sectioned surface of  $4 \times 6 \times 20 \text{ mm}^3$   $\text{Ti}_{34}\text{Zr}_{11}\text{Cu}_{47}\text{Ni}_8$  strip obtained by metal mold casting.

Y.H. LIU, *et. al.*, *Science*, **135**, (2007), 1385-1388.

X.H. LIN, W. JOHNSON, *J. Appl. Phys.*, **78**, (1995), 6514-6519.

# Diffraction

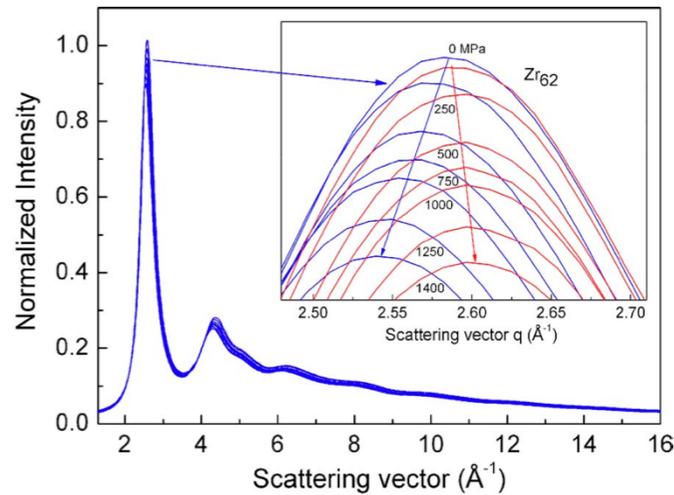


FIG. 1. (Color online) Normalized diffraction curves  $I(q)$  of the tensile direction and the top part magnification of first peaks in  $I(q)$  of tensile/transverse directions changing with increasing stress for  $Zr_{62}Al_8Ni_{13}Cu_{17}$  BMG.

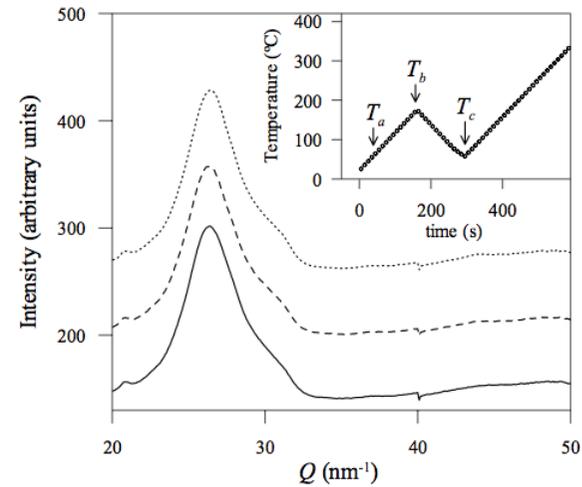


Figure 2.

Main diffraction peak of melt spun  $Al_{90}Fe_5Nd_5$  alloy at 3 points of the annealing protocol detailed in the inset. Solid line:  $T_a=59.8^\circ\text{C}$ . Dashed line:  $T_b=174.6^\circ\text{C}$ . Dotted line:  $T_c=57.8^\circ\text{C}$ .

X.D. WANG, *et. al.*, *Appl. Phys. Lett.*, **91**, (2007), 081913

E. PINEDA, *et. al.*, *J. Alloy Compounds*, **483**, (2009), 578-581.

# Medium Range Structure

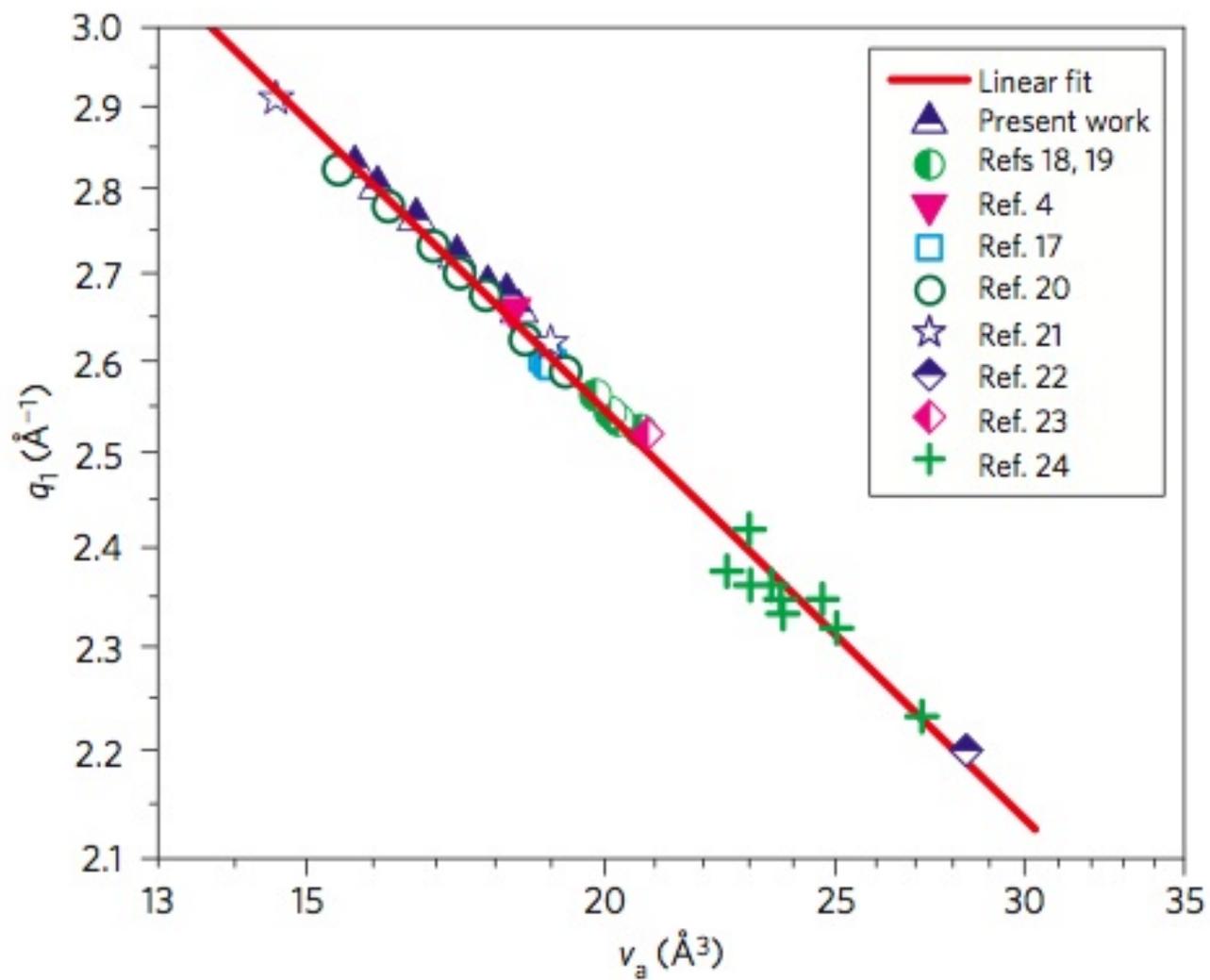
D. MA, A. D. STOICA, X.-L. WANG, Nature Mat., 8, (2009), 30-34.

- The *Medium Range* (MR) structure has been studied using neutron and X-ray diffractions techniques.
- If  $q_1$  denotes the position of the *first sharp diffraction peak* and if  $v$  is the *atomic volume*, the Ehrenfest relation predicts

$$q_1 v^{1/3} = \text{const.} \quad (\text{Ehrenfest})$$

- In BMG's *measurements* show that, at distance of order  $4-6\text{\AA}$ .

$$q_1 v^\alpha = \kappa \quad \alpha = 0.433 \pm 0.007 \quad \kappa = 9.3 \pm 0.2$$



Experimental fit with  
 $q_1 v_a^\alpha = \kappa$

*(D.Ma et al. '09)*

# III - Delone Graphs

# Delone Sets

- The set  $\mathcal{V}$  of atomic positions is *uniformly discrete* if there is  $b > 0$  such that in any ball of radius  $b$  there is at *most* one atomic nucleus.

(Then minimum distance between atoms is  $\geq 2b$ )

- The set  $\mathcal{V}$  is *relatively dense* if there is  $h > 0$  such that in any ball of radius  $h$  there is at *least* one atomic nucleus.

(Then maximal vacancy diameter is  $\leq 2h$ )

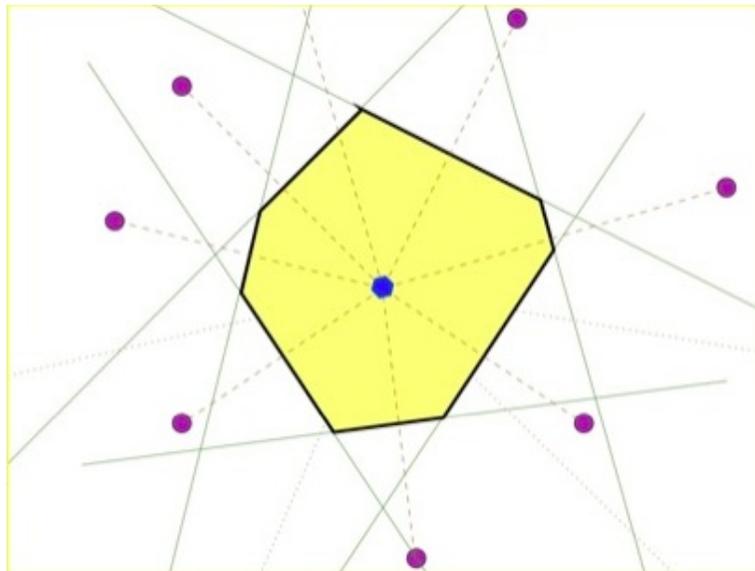
- If  $\mathcal{V}$  is both uniformly discrete and relatively dense, it is called a *Delone set*.
- $\text{Del}_{b,h}$  denotes the set of *Delone sets* with parameters  $b, h$ .

# Voronoi Cells

- Let  $\mathcal{V} \in \text{Del}_{b,h}$ . If  $x \in \mathcal{V}$  its *Voronoi cell* is defined by

$$V(x) = \{y \in \mathbb{R}^d ; |y - x| < |y - x'| \forall x' \in \mathcal{V}, x' \neq x\}$$

$V(x)$  is open. Its closure  $T(x) = \overline{V(x)}$  is called the *Voronoi tile* of  $x$



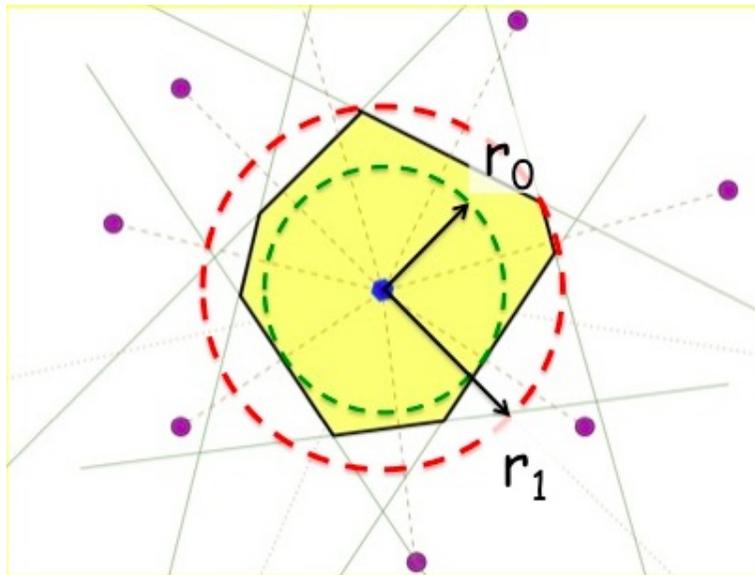
**Proposition:** If  $\mathcal{V} \in \text{Del}_{r_0,r_1}$  the Voronoi tile of any  $x \in \mathcal{V}$  is a convex polytope

# Voronoi Cells

- Let  $\mathcal{V} \in \text{Del}_{b,h}$ . If  $x \in \mathcal{V}$  its *Voronoi cell* is defined by

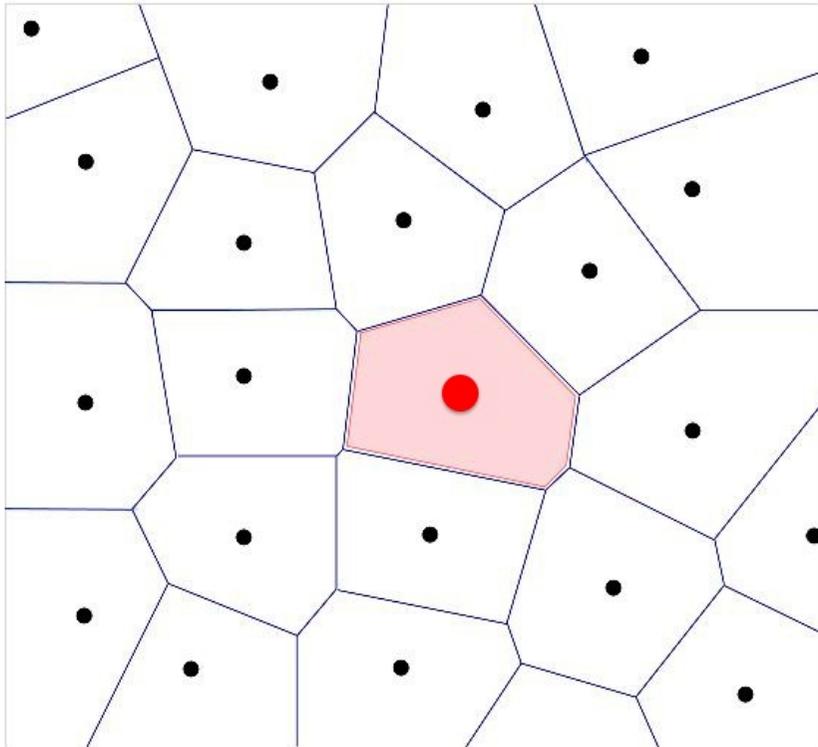
$$V(x) = \{y \in \mathbb{R}^d ; |y - x| < |y - x'| \forall x' \in \mathcal{V}, x' \neq x\}$$

$V(x)$  is open. Its closure  $T(x) = \overline{V(x)}$  is called the *Voronoi tile* of  $x$



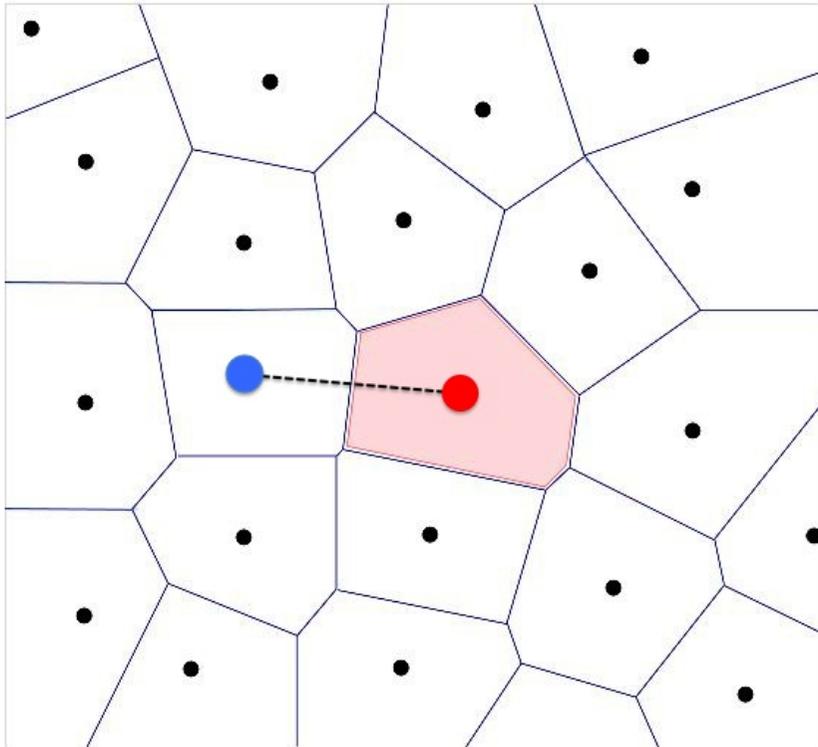
**Proposition:** If  $\mathcal{V} \in \text{Del}_{r_0, r_1}$  the Voronoi tile of any  $x \in \mathcal{V}$  is a convex polytope containing the ball  $\overline{B}(x; r_0)$  and contained in the ball  $\overline{B}(x; r_1)$

# The Delone Graph



**Proposition:** *the Voronoi tiles of a Delone set touch face-to-face*

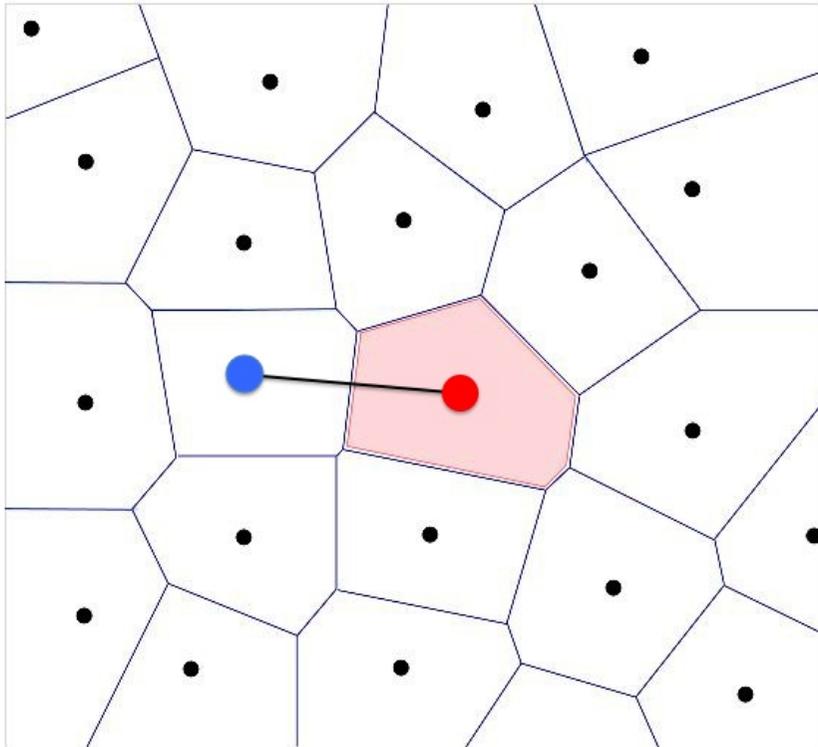
# The Delone Graph



**Proposition:** *the Voronoi tiles of a Delone set touch face-to-face*

Two atoms are *nearest neighbors* if their Voronoi tiles touch along a face of *maximal dimension*.

# The Delone Graph

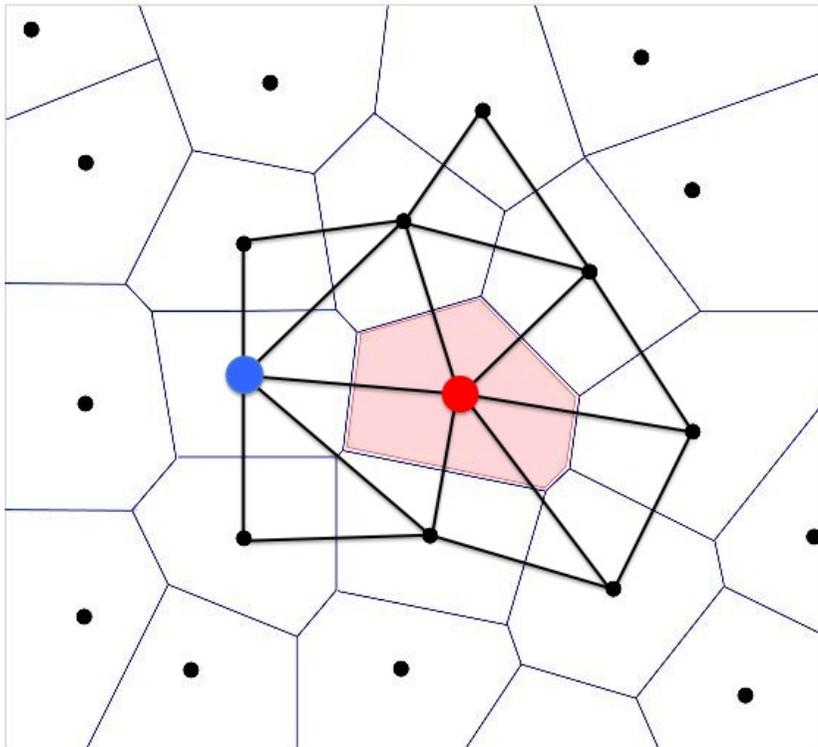


**Proposition:** *the Voronoi tiles of a Delone set touch face-to-face*

Two atoms are *nearest neighbors* if their Voronoi tiles touch along a face of *maximal dimension*.

An *edge* is a pair of nearest neighbors.  $\mathcal{E}$  denotes the set of edges.

# The Delone Graph



**Proposition:** *the Voronoi tiles of a Delone set touch face-to-face*

Two atoms are *nearest neighbors* if their Voronoi tiles touch along a face of *maximal dimension*.

An *edge* is a pair of nearest neighbors.  $\mathcal{E}$  denotes the set of edges.

The family  $\mathcal{G} = (\mathcal{V}, \mathcal{E})$  is the Delone graph.

# The Delone Graph

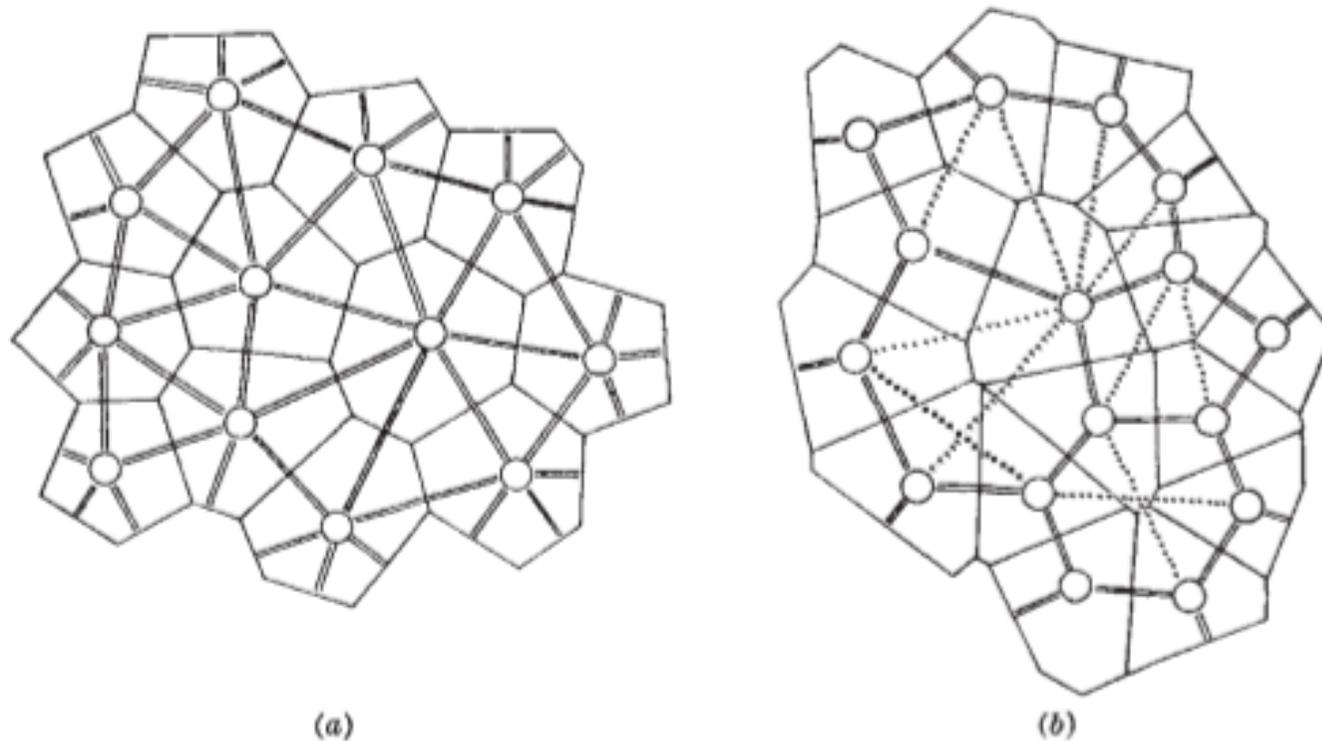


Fig. 1. Diagram of neighbourhood polyhedra, geometrical and physical, for two-dimensional arrays of points. (a) High co-ordinated; —, physical neighbours; (b) low co-ordinated; . . . ., geometrical neighbours

*taken from J. D. BERNAL, Nature, 183, 141-147, (1959)*

# Properties of the D-graph

- **Graphs:** (*non-oriented*) graph  $\mathcal{G} = (\mathcal{V}, \mathcal{E})$  is a pair of sets with a map  $\partial : \mathcal{E} \rightarrow \mathcal{V} \times \mathcal{V}$ , called *boundary map*.
- **Graph maps:**  $f : \mathcal{G} = (\mathcal{V}, \mathcal{E}) \rightarrow \mathcal{G}' = (\mathcal{V}', \mathcal{E}')$  is a pair of maps  $f_v : \mathcal{V} \rightarrow \mathcal{V}'$ ,  $f_e : \mathcal{E} \rightarrow \mathcal{E}'$  such that

$$\partial f_e(e) = (f_v \times f_v) (\partial e)$$

- **Composition:**  $f \circ g = (f_v \circ g_v, f_e \circ g_e)$ .
- **Isomorphism:**  $f : \mathcal{G} = (\mathcal{V}, \mathcal{E}) \rightarrow \mathcal{G}' = (\mathcal{V}', \mathcal{E}')$  is an isomorphism if there is  $g : \mathcal{G}' = (\mathcal{V}', \mathcal{E}') \rightarrow \mathcal{G} = (\mathcal{V}, \mathcal{E})$  with  $f \circ g = \mathbf{1}_{\mathcal{G}'}$ ,  $g \circ f = \mathbf{1}_{\mathcal{G}}$ .

# Properties of the D-graph

**Proposition:** *Given two integers  $N_v, N_e$ , the number of graphs modulo isomorphism with less than  $N_v$  vertices and less than  $N_e$  edges is finite*

**Consequence:** There are only finitely many D-Graphs representing a configuration of the glass in a ball of finite radius. D-graphs discretize the information.

# Properties of the D-graph

- The incidence number  $n_x$  of a vertex  $x \in \mathcal{V}$  is bounded by

$$d + 1 \leq n_x \leq \frac{\sqrt{\pi} \Gamma\{(d - 1)/2\}}{\Gamma(d/2) \int_0^{\theta_m} \sin^{d-1}(\theta) d\theta}, \quad \sin \theta_m = b/2h.$$

- A *local patch* of radius  $n \in \mathbb{N}$  is an *isomorphism class* of subgraphs  $(x, \mathcal{V}_x, \mathcal{E}_x)$  of the Delone graph, such that  $x \in \mathcal{V}$ ,  $\mathcal{V}_x$  is the set of vertices at graph-distance at most  $n$  from  $x$ .
- If  $\mathcal{P}_n$  denote the *set of local patches* of radius  $n$  then there is  $C = C(b, h) > 0$  such that

$$\#\mathcal{P}_n \leq e^{C(2n+1)^d}$$

# Likelyhood: Genericity

*Genericity* is a topological concept.

- In a topological space  $X$ , a subset  $A \subset X$  is *dense* if any nonempty open set intersects  $A$ .
- A  $G_\delta$ -set is the intersection of a countable family of open sets.
- *Baire Category Theorem*: if  $X$  is homeomorphic to a complete metric space, then a countable intersection of dense open sets is dense.
- A property is called *generic* when it holds in a dense  $G_\delta$ .

## Likelihood: Almost Surely

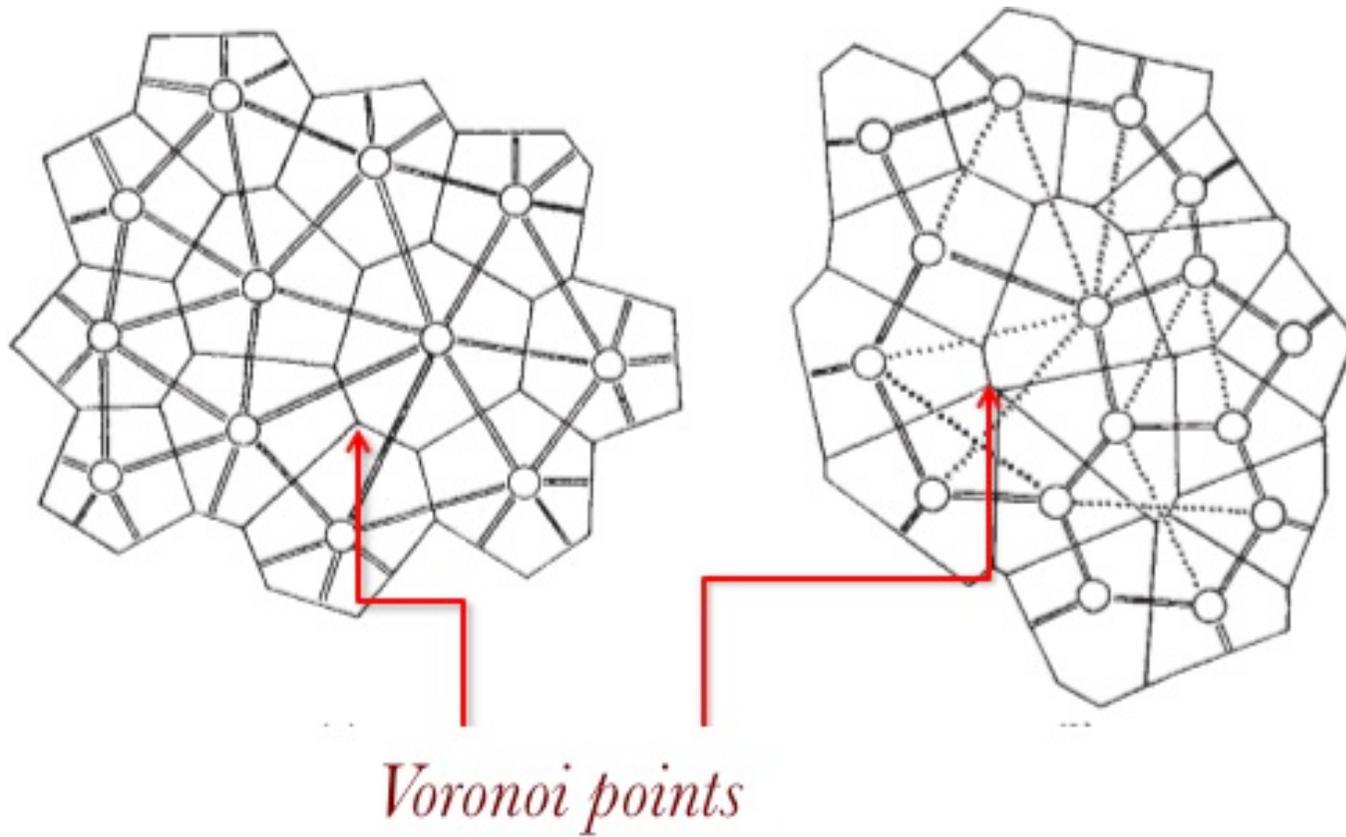
*Almost Surely* is a measure theoretic or probability concept concept.

- A *probability space* is a triple  $(X, \Sigma, \mathbb{P})$ , where  $X$  is a set,  $\Sigma$  a family of subsets of  $X$  containing  $X$ , that is closed under complementation and countable intersection (called the  *$\sigma$ -algebra of measurable sets*) and  $\mathbb{P}$  is a *probability measure*, namely  $\mathbb{P} : \Sigma \rightarrow [0, 1]$  satisfying standard assumptions.
- In a probability space  $(X, \Sigma, \mathbb{P})$ , a property is *almost sure* whenever it occurs in a measurable subset  $A \in \Sigma$  having probability  $\mathbb{P}(A) = 1$ .

# Likelihood

- There are examples of *generic subsets* of  $[0, 1]$  with *zero probability* (w.r.t the Lebesgue measure), the complement of which is almost sure without being generic.
- If  $X \subset \mathbb{R}^n$  is closed and if  $\mathbb{P} = F(x)d^n x$  is “absolutely continuous”, then a property valid of a dense open set  $U \subset X$ , with piecewise smooth boundary, is both generic and almost sure.

# Voronoi Points

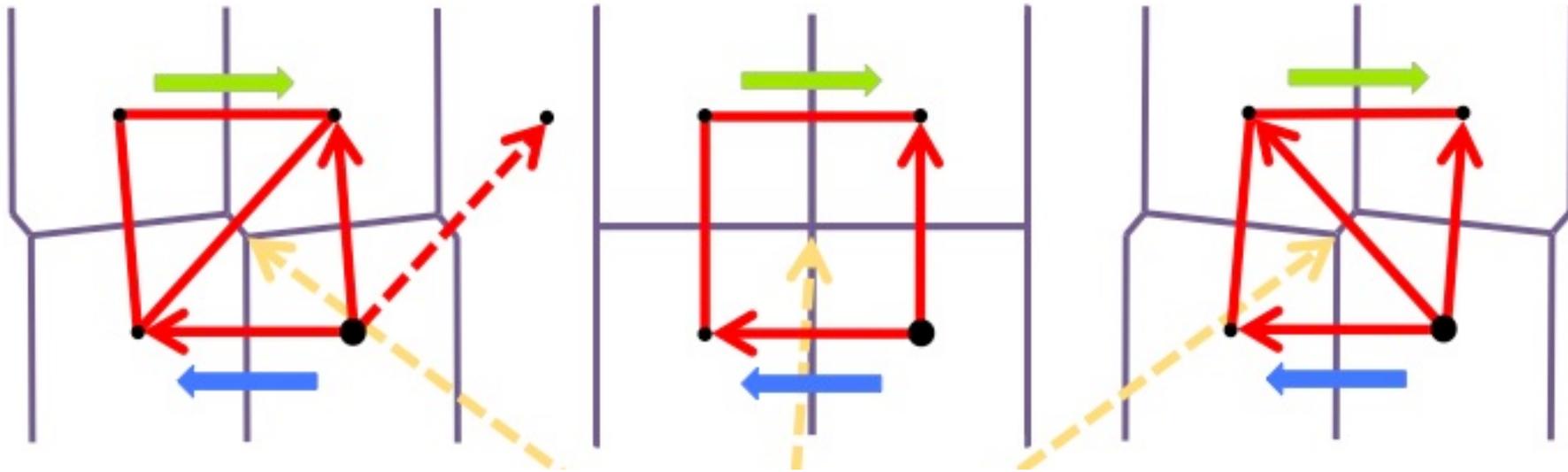


The vertices of the Voronoi cells are called *Voronoi Points*.

# Voronoi Points

- **Theorem:** *A Voronoi point is at equal distance from every atom the Voronoi tile of which it belongs.*
- **Theorem:** *Generically and almost surely a Voronoi point belongs to exactly  $d + 1$  Voronoi tiles in dimension  $d$ .*

## Generic Local Patches

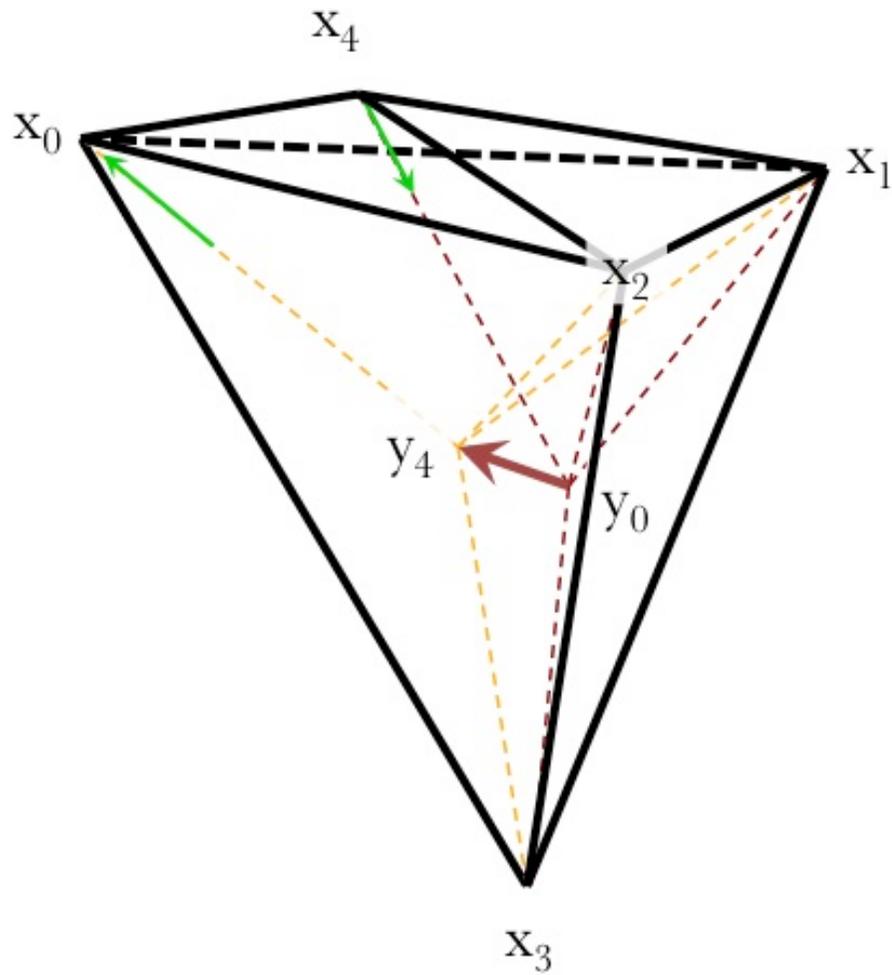


Shear modifies local patches. The middle one is *unstable*. The transition from left to right requires transiting through a *saddle point* of the potential energy.

*The Voronoi cell boundaries are shown in blue.*

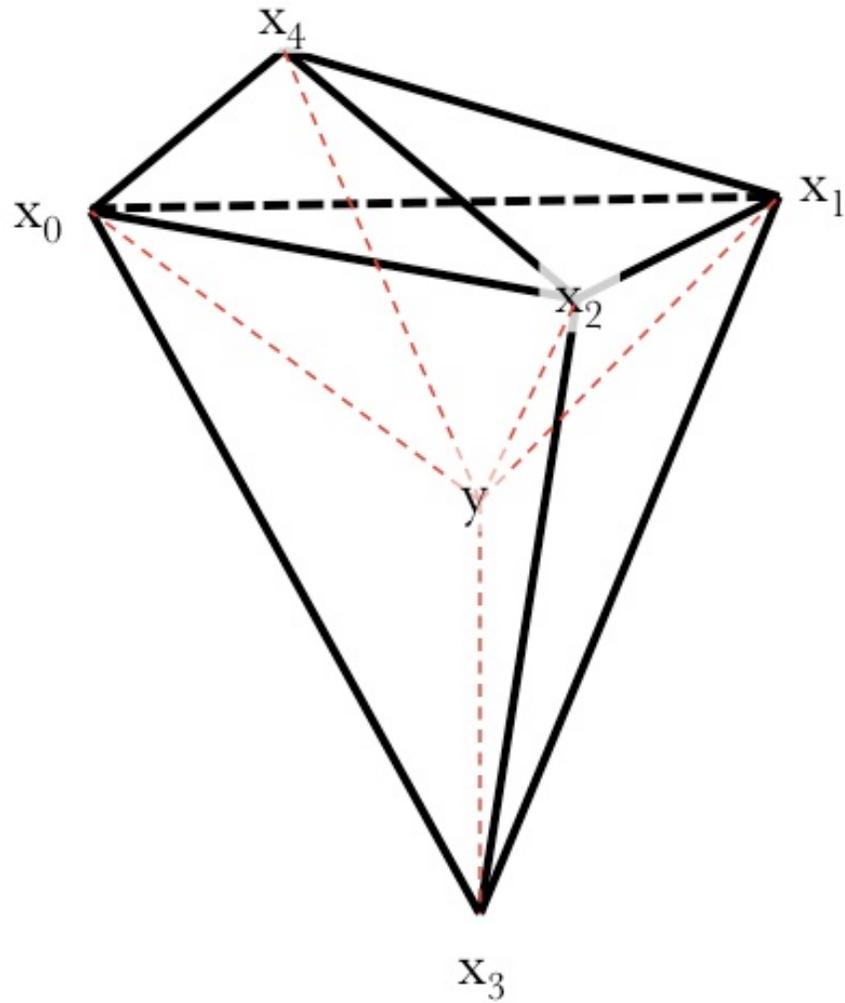
*At the bifurcation a Voronoi vertex touches one more Voronoi cell than in the generic case*

# Generic Local Patches



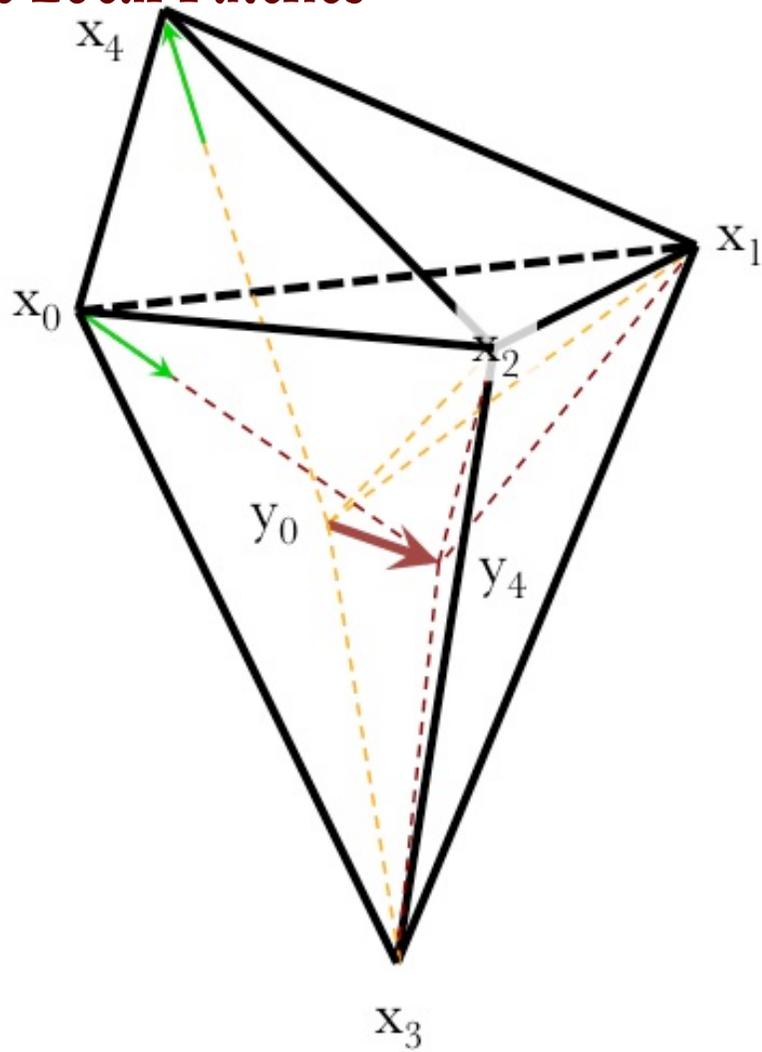
*An example of a generic 3D bifurcation.*

# Generic Local Patches



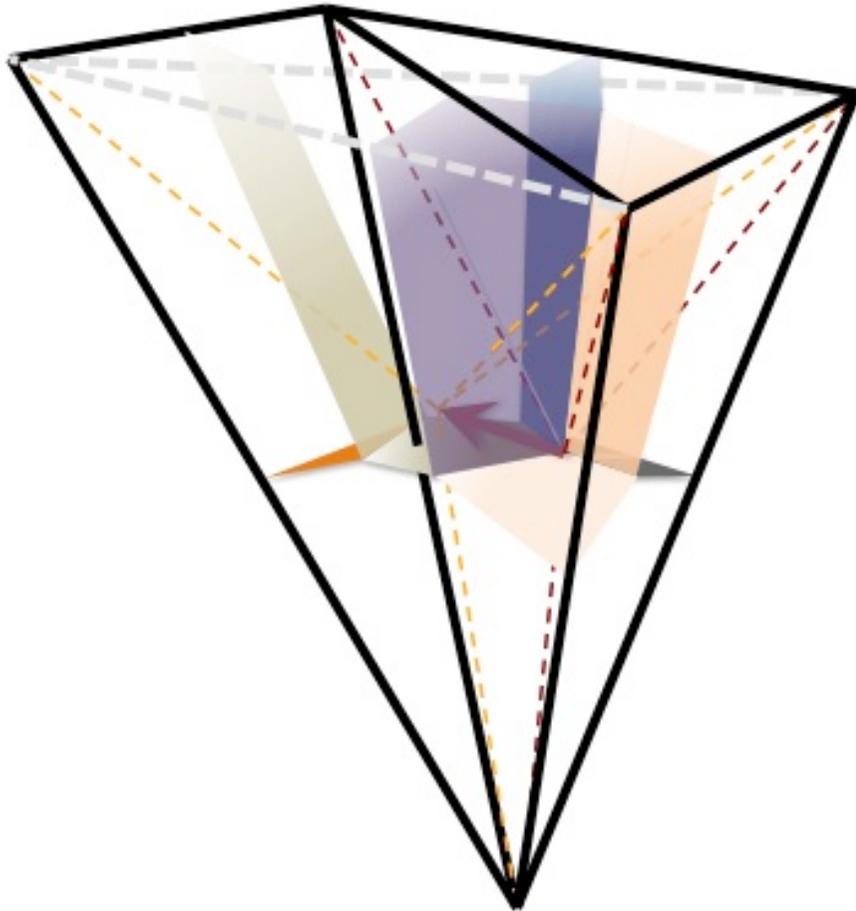
*An example of a generic 3D bifurcation.*

# Generic Local Patches



*An example of a generic 3D bifurcation.*

# Generic Local Patches

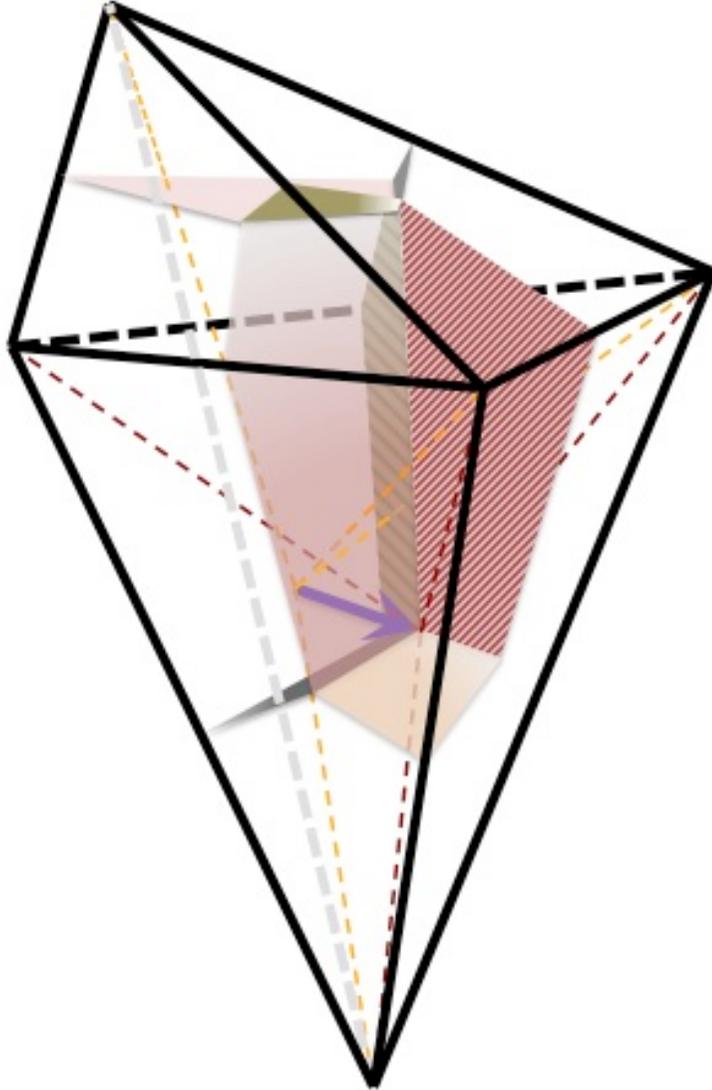


*An example of a generic 3D bifurcation.*

*Graph changes*

- The graph edges are indicated in black.
- The grey dotted edges have disappeared during the bifurcation.
- The colored plates are the boundaries of the Voronoi cells.

# Generic Local Patches



*An example of a generic 3D bifurcation.*

*Graph changes*

- The graph edges are indicated in black.
- The grey dotted edges have disappeared during the bifurcation.
- The colored plates are the boundaries of the Voronoi cells.

# Acceptance Domains

- Given a local patch  $\mathcal{G} \in \mathcal{P}_n$  its acceptance domain  $\Sigma_{\mathcal{G}}$  is the set of all atomic configurations  $\mathcal{V} \in \text{Del}_{b,h}$  having  $\mathcal{G}$  as their *local patch around the origin*.
- A local patch is *generic* whenever a small local deformation of the atomic configuration does not change the corresponding graph. Let  $\mathfrak{B}_n \subset \mathcal{P}_n$  be the set of *generic local patches* of radius  $n$ .
- **Theorem:**  $\mathcal{G} \in \mathfrak{B}_n$  if and only if  $\Sigma_{\mathcal{G}}$  is open and its boundary is *piecewise smooth*.  
*The union of acceptance domains of the generic patches of size  $n$  is dense.*  
*In particular almost surely and generically an atomic configuration admits a generic local patch.*

# Contiguosness

- The *boundary* of the acceptance domain of a generic graph contains a relatively open dense subset of codimension 1.
- **Definition:** *two generic graphs  $\mathcal{G}, \mathcal{G}' \in \mathcal{Q}_n$  are contiguous whenever their boundary share a piece of codimension one.*
- The set  $\mathcal{B}_n$  itself can then be seen as the set of vertices of a graph

$$\mathcal{G}_n = (\mathcal{B}_n, \mathcal{E}_n)$$

called the *graph of contiguousness* where *an edge  $E \in \mathcal{E}_n$  is a pair of contiguous generic local patches.*

## IV - The Anankeon Theory

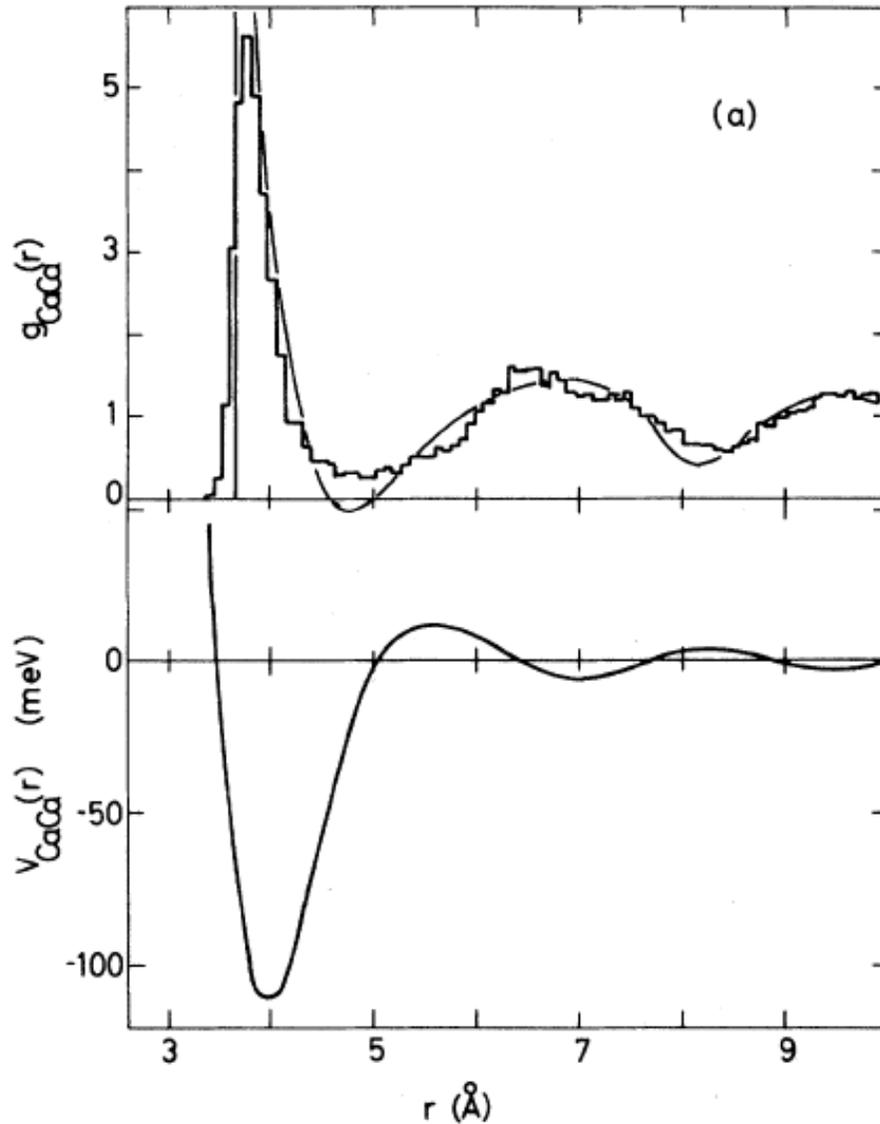
# Fundamental Laws

1. The *Coulomb forces* between atomic cores and valence electrons create *chemical bonding* and *cohesion* in solids
2. Electrons are *fermions*: they resist compression. For free Fermi gas ( $\ell_{e-e}$  = average  $e - e$  distance,  $P$ =pressure)

$$P \sim \ell_{e-e}^5$$

3. In metals, valence electrons are *delocalized*, approximately free. Atomic cores localize themselves to *minimize* the Fermi sea *energy* (*jellium*).
4. A good description of the effective atom-atom interactions is provided by *pair potential* with strong repulsion at short distances, Friedel's oscillations at medium range and exponentially decaying tail.

# Pair Potentials



An example of atom-atom pair potential in the metallic glass  $\text{Ca}_{70}\text{Mg}_{30}$

*Top:* the pair creation function  
*Bottom:* the graph of the pair potential

*taken from*  
*J. HAFNER, Phys. Rev. B, 27, 678-695 (1983)*

# Dense Packing

1. The shape of the pair potential suggests that there is a *minimal distance* between two atoms.
2. Liquid and solids are *densely packed*. This suggests that there is a *maximal size for vacancies*.
3. However, Mathematics (*ergodic theory*) implies that, given an  $\epsilon > 0$ , with probability one
  - there are pairs of atoms with distance less than  $\epsilon$
  - there are vacancies with radius larger than  $1/\epsilon$
4. But these rare events are not seen in practice because their *lifetime is negligibly small* (Bennett et al. '79).

# Persistence

- *Persistence* theory gives an idea about why large vacancies have a short lifetime. On discrete subset  $\mathcal{V} \subset \mathbb{R}^d$ , let,  $(n_x)_{x \in \mathcal{V}}$ , be a family of *i.i.d random variables* with  $n_x \in \{0, 1\}$  and  $\text{Prob}\{n_x = 0\} = p > 0$ ,  $\text{Prob}\{n_x = 1\} = 1 - p > 0$ .
- Then, if  $\Lambda \subset \mathcal{V}$  is a finite set, let  $P_\Lambda(t)$  be the probability that  $n_x = 0$  for  $x \in \Lambda$  and times between 0 and  $t$ , given that  $n_x = 0$  at  $t = 0$  for  $x \in \Lambda$ . By independence

$$P_\Lambda(t) = \prod_{x \in \Lambda} P_{\{x\}}(t)$$

- Usually  $P_{\{x\}}(t) \simeq e^{-t/\tau}$ . Hence the life time of  $\Lambda$  as a vacancy is  $\tau/N$  if  $\Lambda$  has  $N$  atoms.

# Bonds and Phonons

T. EGAMI, *Atomic Level Stress*, Prog. Mat. Sci., **56**, (2011), 637-653.

1. Atoms can be related by *edges* using Voronoi cells construction. Long edges are *loose*. Short edges are *bonds*.
2. If  $r$  is the vector linking two atoms of a bond, there is a local  $6D$  *stress tensor* defined by

$$\sigma^{\alpha\beta} = V'(|r|) \frac{r^\alpha r^\beta}{|r|}$$

3. **Liquid Phase:** Bonds constitute the *dominant* degrees of freedom ! Phonons are *damped*.
4. **Glass Phase:** Phonons are the *dominant* degrees of freedom. Bonds are *blocked*.

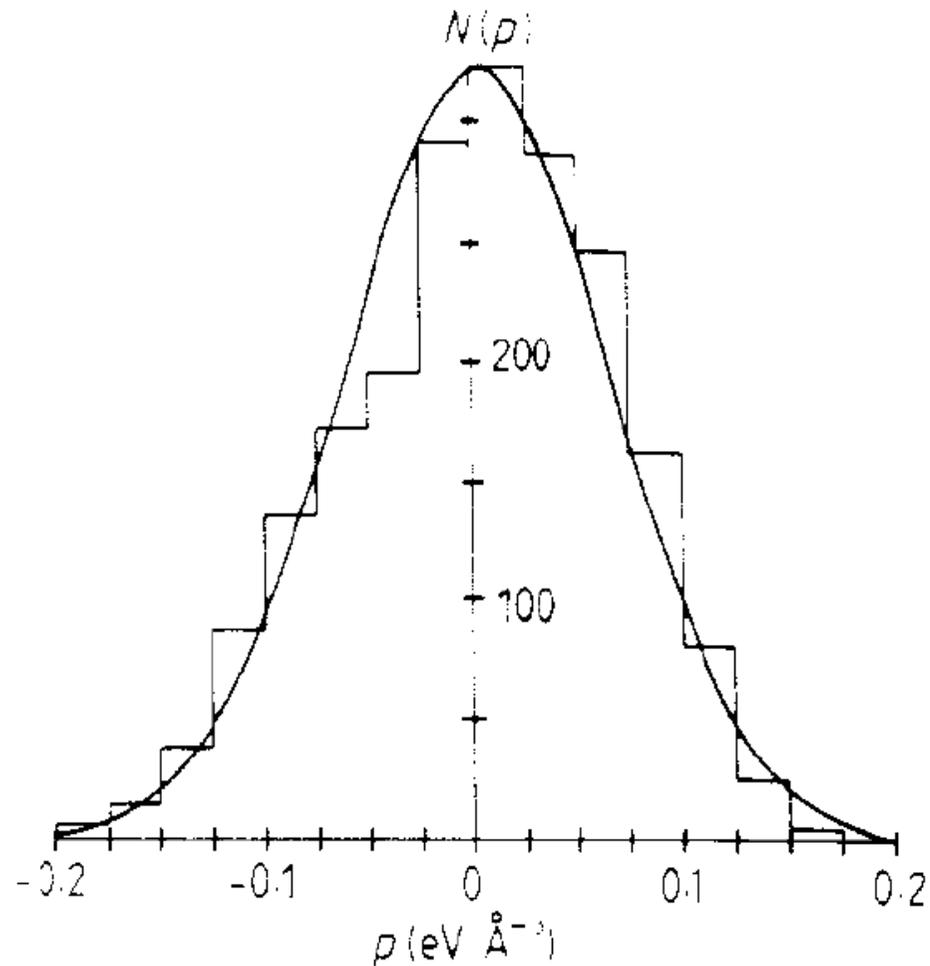
# Local Stress Distribution

1. In the liquid state atoms do not find a position minimizing the potential energy, due to *geometrical frustration*. Thermal agitation result in atomic *bond exchanges*, to help atoms minimize their potential energy.
2. The stress tensors associated with bonds behave like *independent random Gaussian variables* !
3. Thanks to *isotropy*, this can be seen on the local *pressure*  $p$  and the von Mises local *shear stress*  $\tau$

$$p = \frac{1}{3} \sum_{\alpha} \sigma^{\alpha\alpha}$$

$$\tau = \sqrt{\sum_{\alpha < \beta} |\sigma^{\alpha\beta}|^2}$$

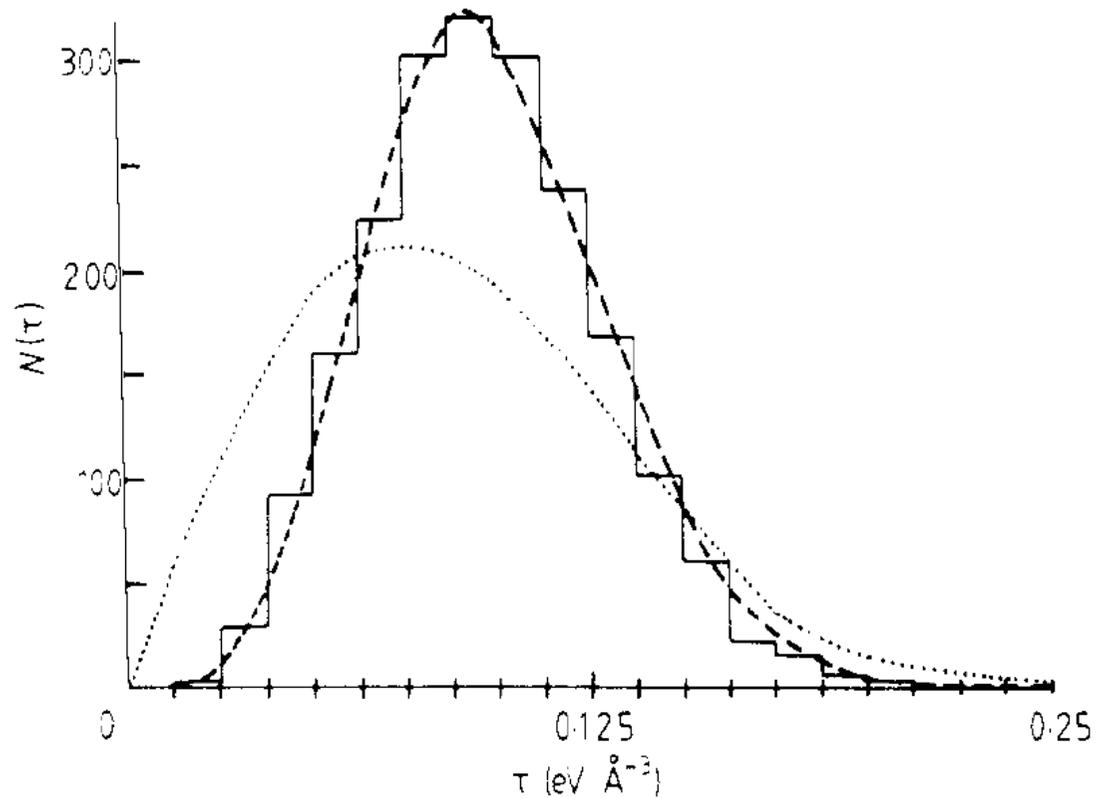
# Local Stress Distribution



Pressure distribution in  
amorphous and liquid metals

*taken from*  
T. EGAMI & D. SROLOVITZ, *J. Phys. F*, **12**,  
2141-2163 (1982)

# Local Stress Distribution



Shear stress distribution in  
amorphous and liquid  
metals

*Dotted curve: 2D-Gaussian*

*Broken curve: 5D-Gaussian*

*taken from*

*T. EGAMI & D. SROLOVITZ, J. Phys. F, 12,  
2141-2163 (1982)*

# The Anankeon Theory

*The bond degrees of freedom are the response of atoms to the **stressful** situation in which they are trying to find a better comfortable position, in vain.*

# The Anankeon Theory

*The bond degrees of freedom are the response of atoms to the **stressful** situation in which they are trying to find a better comfortable position, in vain.*

*Behind the previous concept of bond degrees of freedom, there is the notion of **stress, constraint, necessity, unrest** and even torture.*

# The Anankeon Theory

*The bond degrees of freedom are the response of atoms to the **stressful** situation in which they are trying to find a better comfortable position, in vain.*

*Behind the previous concept of bond degrees of freedom, there is the notion of **stress, constraint, necessity, unrest** and even torture.*

*There is a character of the Greek mythology that could fit with this concept:*

**the goddess Ananke**

*whose name comes from the greek word **anagkeia** meaning the **stress of circumstances**.*

# The Anankeon Theory

*The bond degrees of freedom are the response of atoms to the **stressful** situation in which they are trying to find a better comfortable position, in vain.*

*Behind the previous concept of bond degrees of freedom, there is the notion of **stress, constraint, necessity, unrest** and even torture.*

*There is a character of the Greek mythology that could fit with this concept:*

## **the goddess Ananke**

*whose name comes from the greek word **anagkeia** meaning the **stress of circumstances**. Ananke was representing a power above all including the Gods of the Olympe "even gods don't fight against Ananke" claims a scholar. This character presided to the **creation of the world**, in various versions of the Greek mythology. It expresses the concepts of "**force, constraint, necessity**" and from there it also means "**fate, destiny**" to lead to the concepts of compulsion, torture.*(from Wikipedia)**

# The Anankeon Theory

For this reason the configurational degrees of freedom associated with the stress tensor on each bond will be called

**ANANKEONS**

# The Anankeon Theory

- Edge partition function

$$Z(e) = (1 - \pi) \int_{\mathbb{R}^6} e^{-(p^2/2B + \tau^2/2G)/k_B T} dp d^5\tau + \pi$$

- $\pi$  is the probability for an edge to be *loose*  
( $V_0$  is the pair-potential *maximal depth*)

$$\pi \sim e^{-V_0/k_B T}$$

- $B$  is the *bulk modulus*
- $G$  is the *shear modulus*
- Edge free energy  $F(e) = -k_B T \ln Z(e)$

## The Anankeon Theory

As a consequence of the anankeon theory, at high temperature, the total potential energy per edge,  $3/2 k_B T$ , is equally distributed over the six elastic self-energy of the stress components (equipartition)

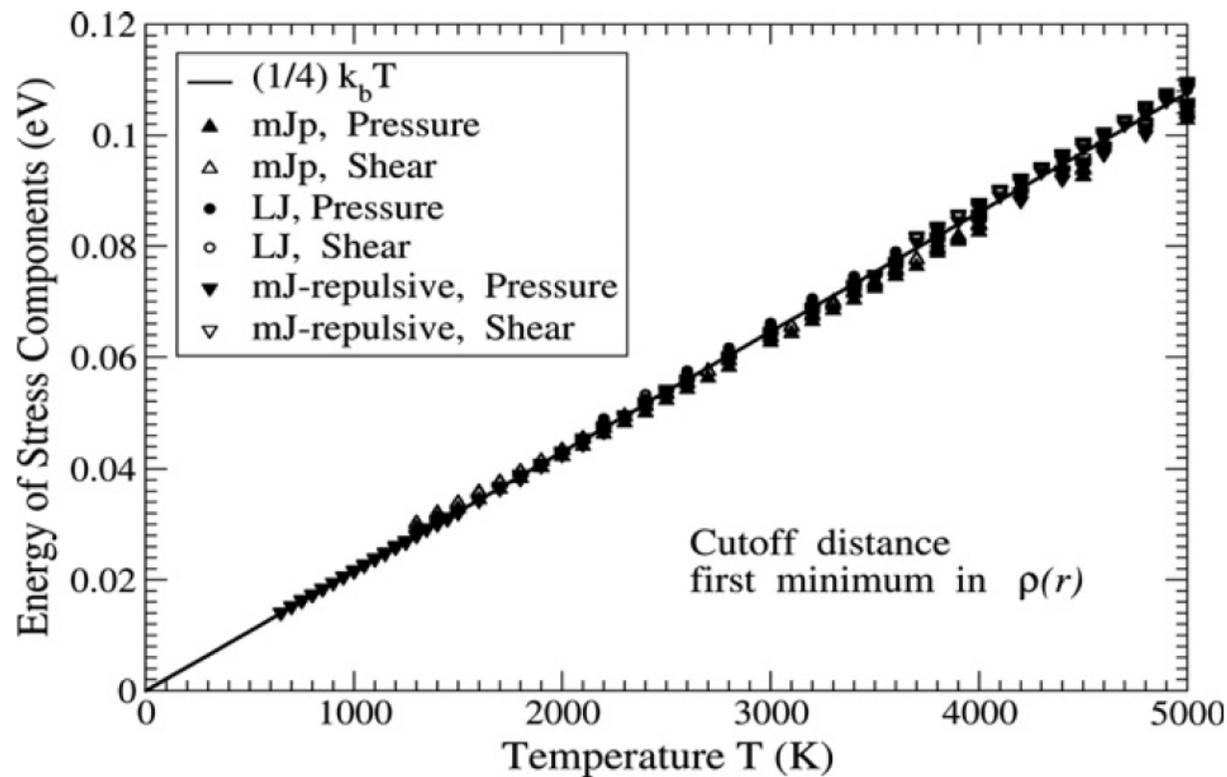
$$\frac{\langle p^2 \rangle}{2B} = \frac{\langle \tau^2 \rangle}{2G} = \frac{k_B T}{4}$$

In particular, the specific heat follows a *law of Dulong-Petit*

$$C_p \stackrel{T \uparrow \infty}{\sim} \frac{3k_B}{2}$$

The corresponding degrees of freedom are the 6 components of the *stress tensor* on each bond.

# The Anankeon Theory



Elastic self-energy of  
atomic level stresses  
tested for various pair  
potentials

*LJ*: Lennard-Jones potential  
*mJ*: Johnson potential including  
Friedel's oscillations.

*taken from*  
S.-P. CHEN, T. EGAMI & V. VITEK,  
*Phys. Rev. B*, **37**, 2440-2449, (1988)

# The Anankeon Theory

As the temperature decreases, the local edges feel a *long-range stress field* around them. This field can be described through a mean field theory using *continuum elasticity* (Eshleby '57). The stress field is renormalized as

$$K_\alpha \frac{\langle p^2 \rangle}{2B} = K_\gamma \frac{\langle \tau^2 \rangle}{2G} = \frac{k_B T}{4} \quad K_\alpha = \frac{3(1-\nu)}{2(1-2\nu)} \quad K_\gamma = \frac{15(1-\nu)}{7-5\nu}$$

with  $\nu = \text{Poisson ratio}$ . This leads to a prediction of the *glass transition temperature* where  $\epsilon_\nu^{T,crit} \simeq 0.095$  is the *critical strain* computed from percolation theory (Egami T, Poon SJ, Zhang Z, Keppens V., '07)

$$T_g = \frac{2BV}{k_B K_\alpha} (\epsilon_\nu^{T,crit})^2$$

# V - Towards a Dissipative Dynamics

(WORK IN PROGRESS)

# Configuration Space

- Given  $\mathcal{G} \in \mathfrak{B}_n$ , each edge  $e$  of  $\mathcal{G}$  is either *loose* or a *bond*. This can be represented by a random variable  $N_e \in \{0, 1\}$  where
  - $N_e = 0$  if  $e$  is *loose*
  - $N_e = 1$  if  $e$  is a *bond*
  - $\text{Prob}\{N = 0\} = \pi$ ,  $\text{Prob}\{N = 1\} = 1 - \pi$
  - if  $e \neq e'$ , then  $N_e, N_{e'}$  are *independent*.
- Each edge  $e \in \mathcal{G}$  with  $N_e = 1$  supports the *six components* of a local stress tensor  $\sigma_e$  which is distributed according to *Maxwell-Boltzmann*

$$\text{Prob} \left\{ \sigma_e \in \Delta \subset \mathbb{R}^6 \mid N_e = 1 \right\} = \int_{\Delta} \exp \left\{ - \left( \frac{p_e^2}{2Bk_B T} + \frac{\tau_e^2}{2Gk_B T} \right) \right\} d^6 \sigma_e$$

# Interactions

- To build the Hamiltonian describing both the dynamics and the equilibrium, the *continuum elasticity theory* of Eshleby (*Eshleby '57*) must be included as a mean field theory interaction between bonds.
- To describe both the liquid and the solid phase, *phonons* must be included. Since phonons are *damped* in the liquid phase this has to be included in the interaction scheme.
- *Percolation* theory applies thanks to the *loose-bond dichotomy* in the anankeon theory.

# Markov Dynamics

- The contiguosness graph  $\mathfrak{G}_n$  should leads to a *Markov process* represented by the rate probability of transition  $\mathbb{P}_{\mathfrak{G} \rightarrow \mathfrak{G}'}^n$  between two generic *contiguous* local patches

$$\mathbb{P}_{\mathfrak{G} \rightarrow \mathfrak{G}'}^n = \Gamma(\mathfrak{G} \rightarrow \mathfrak{G}') \exp \left\{ - \left( F_{\mathfrak{G}'}(\sigma') - F_{\mathfrak{G}}(\sigma) \right) / k_B T \right\}$$

where  $F_{\mathfrak{G}}(\sigma)$  represents the configuration dependent free energy associated with the local patch  $\mathfrak{G} \in \mathfrak{B}_n$ .

- Here  $\Gamma(\mathfrak{G} \rightarrow \mathfrak{G}') \sim e^{-W/k_B T}$  is proportional to the inverse of the *typical transition time*. This time is controlled by the height  $W$  of the potential energy barrier between the two configurations, following an *Arrhenius law*.

# Markov Dynamics

- Once the model established the *infinite volume limit*, corresponding to the limit  $n \rightarrow \infty$  must be considered. Standard theorems exist in the literature on *Dirichlet forms* about the existence and the uniqueness of such limiting processes.
- Then it will be necessary to prove that, within this model, the main properties discovered by theoreticians are actually a consequence of the model.
- One critical data will be to look at the time scale involved in the liquid and the glassy state.



Thanks for Listening!